# Optical Properties of Solids: Lecture 12


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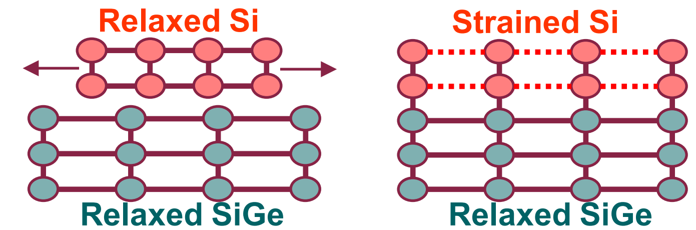
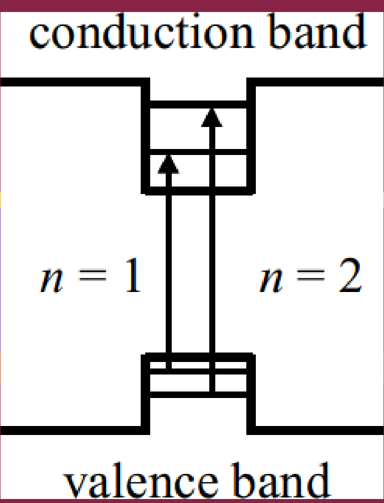
# Optical Properties of Solids: Lecture 12

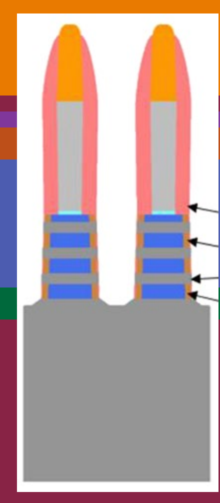
Quantum structures: wells, wires, dots
Absorption and emission of quantum wells
Intersubband absorption
Quantum cascade lasers

Defects (deep, shallow)

Stress and strain





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# References: Band Structure and Optical Properties

## Solid-State Theory and Semiconductor Band Structures:

- Mark Fox, Optical Properties of Solids (6,8,9)
- Ashcroft and Mermin, Solid-State Physics
- Yu and Cardona, Fundamentals of Semiconductors
- Dresselhaus/Dresselhaus/Cronin/Gomes, Solid State Properties
- Cohen and Chelikowsky, Electronic Structure and Optical Properties
- Klingshirn, Semiconductor Optics
- Grundmann, Physics of Semiconductors
- Ioffe Institute web site: NSM Archive
  http://www.ioffe.ru/SVA/NSM/Semicond/index.html

# Outline

Quantum confinement and Heisenberg uncertainty principle
Growth of quantum structures
**Carbon nanostructures, two-dimensional materials**
**Electronic states, quantum well absorption and emission**
**Intersubband transitions**
**Metamaterials and metasurfaces**

**Defects**
**Transition metal and rare earth impurities in insulators**
**Shallow defects in semiconductors**

**Stress and strain, deformation potentials**



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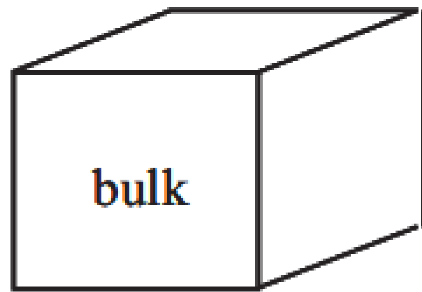
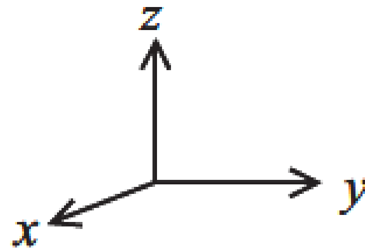
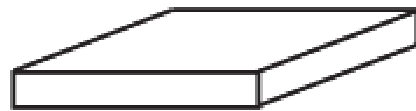
# Quantum structures and density of states

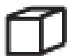


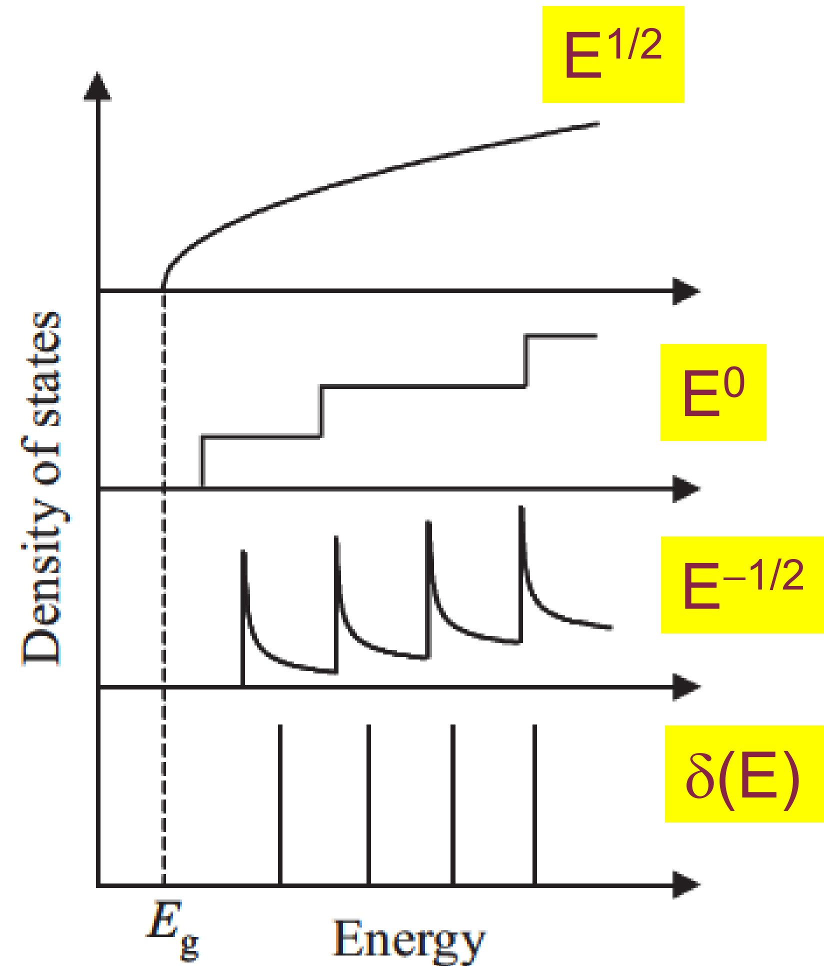



quantum well

quantum wire



quantum dot

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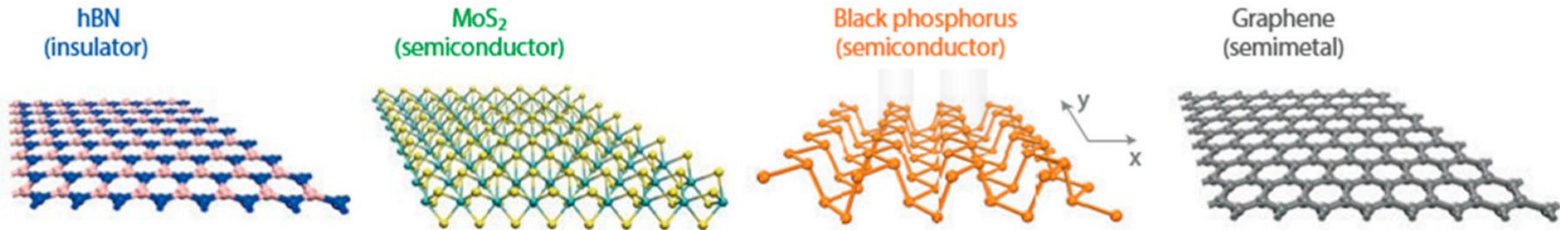
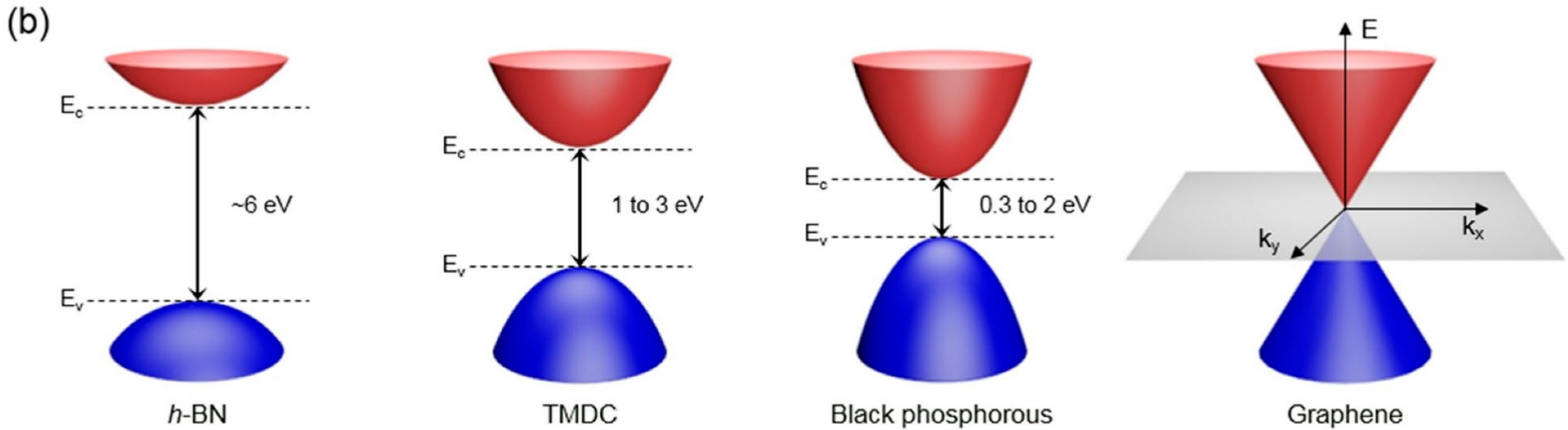
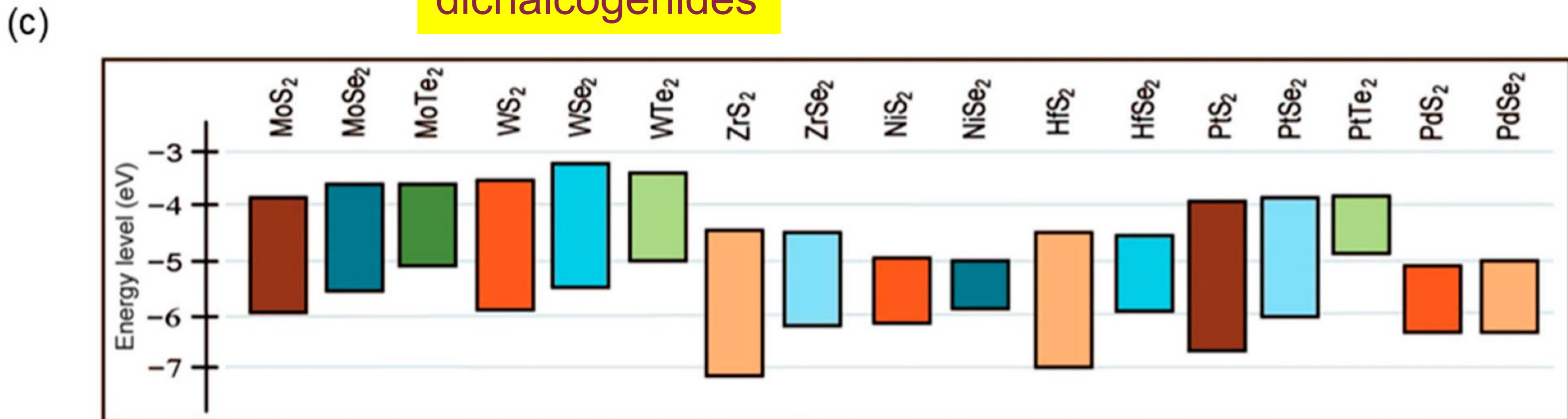
# Two-dimensional semiconductors

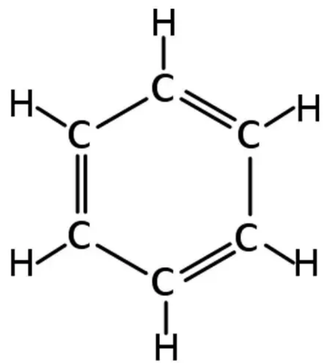


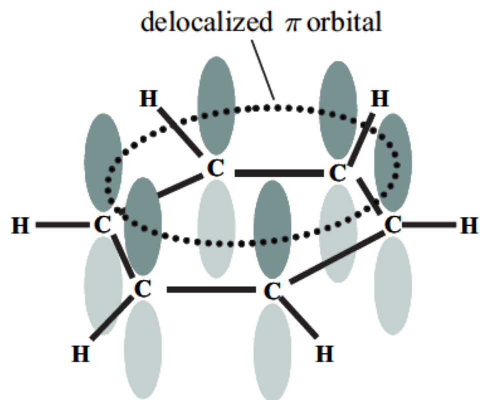



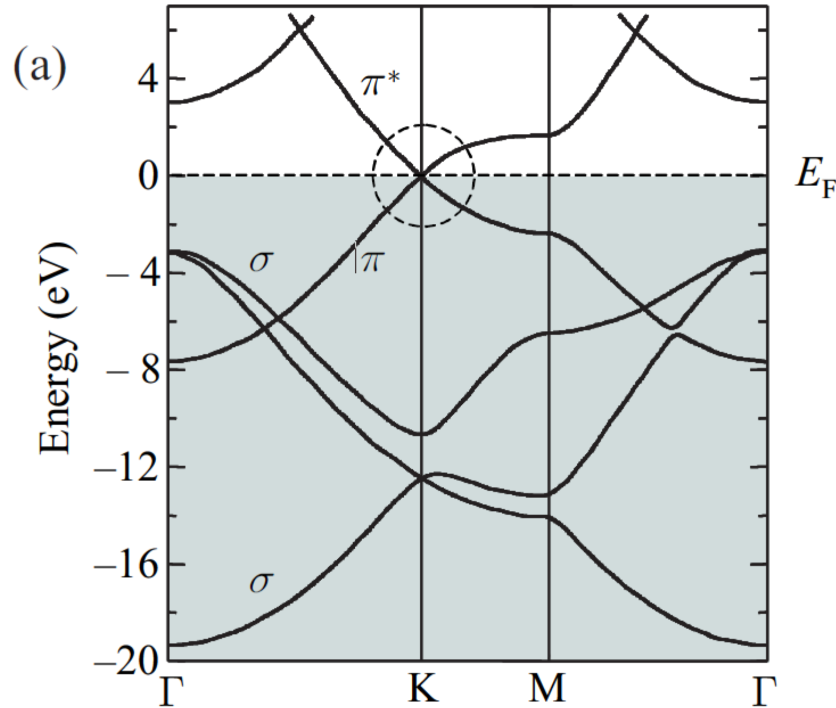

dichalcogenides

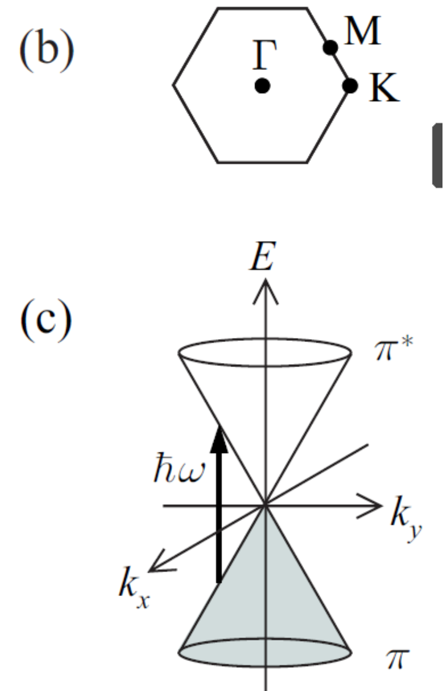


# Dirac point in graphene




Graphite: Strong bond in the graphene plane, weak bond between planes. Remove single graphene plane.

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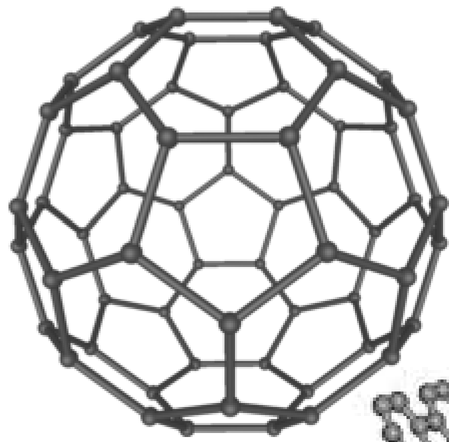
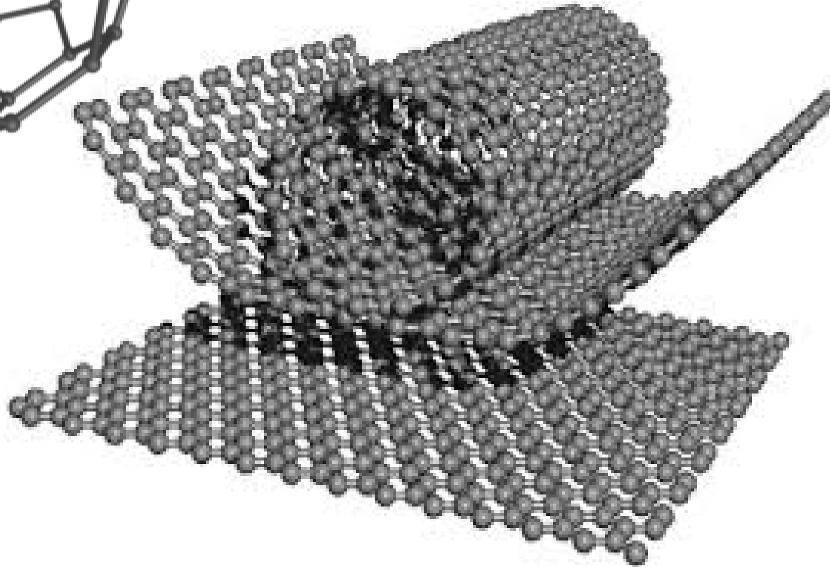
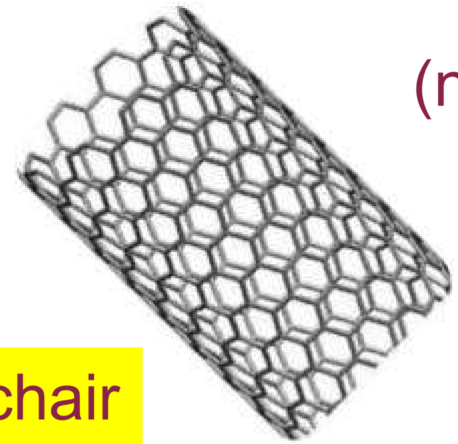
# Carbon nanotubes and fullerenes

(n,m)

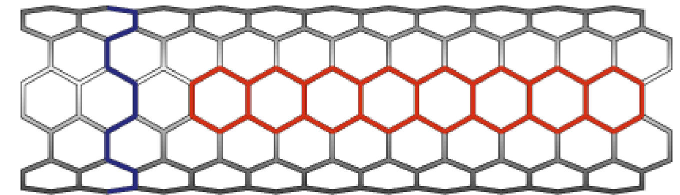


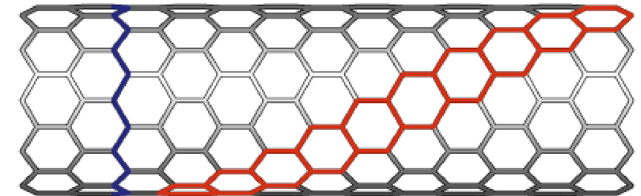
C-60





armchair




Zig-zag

Graphene sheets are rolled up to form tubes.
Many Raman studies, doping with defects.

Jorio, Dresselhaus, Dresselhaus
Carbon Nanotubes

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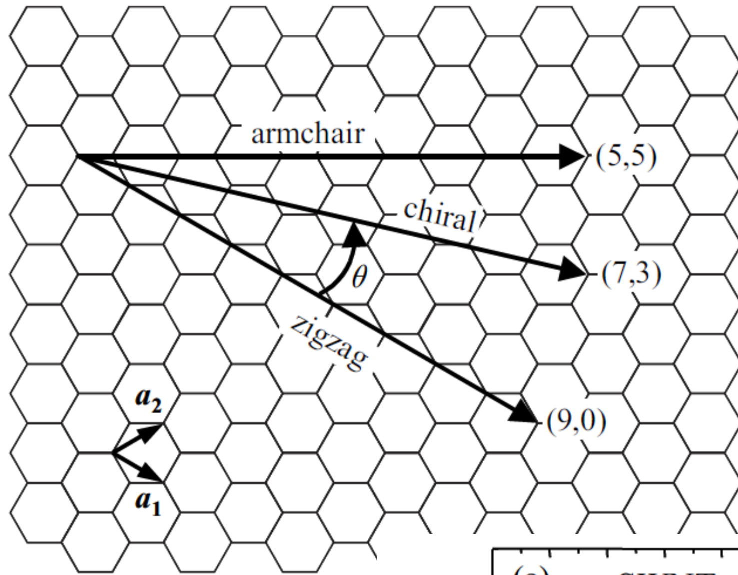
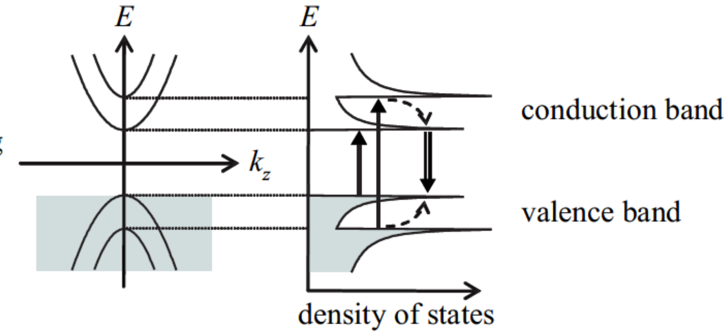
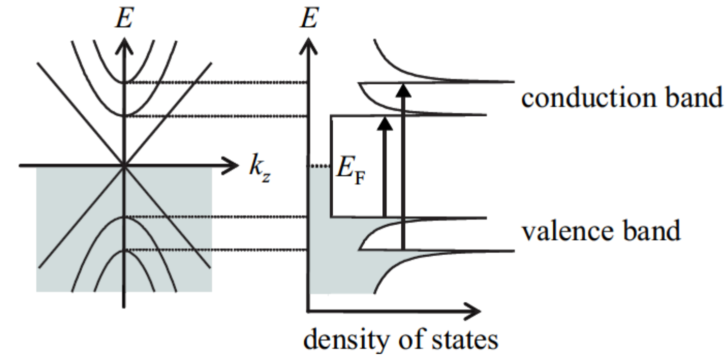
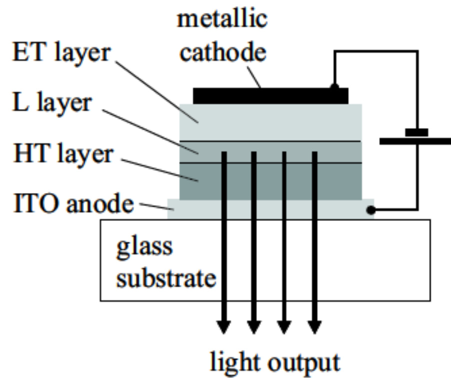
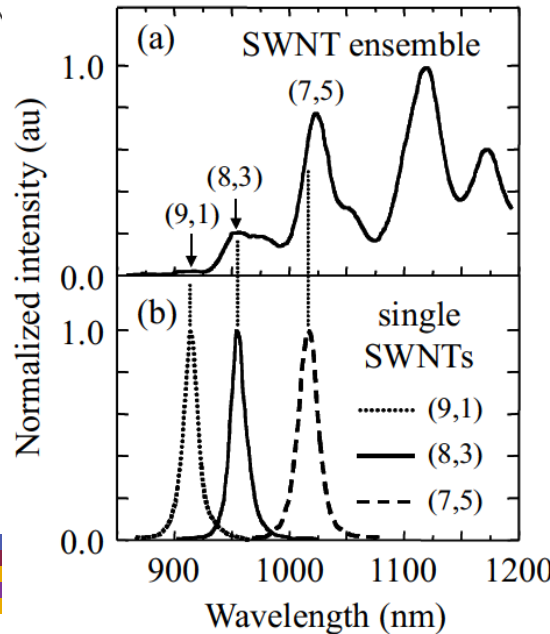
# (n,m) Carbon nanotubes: electrical properties

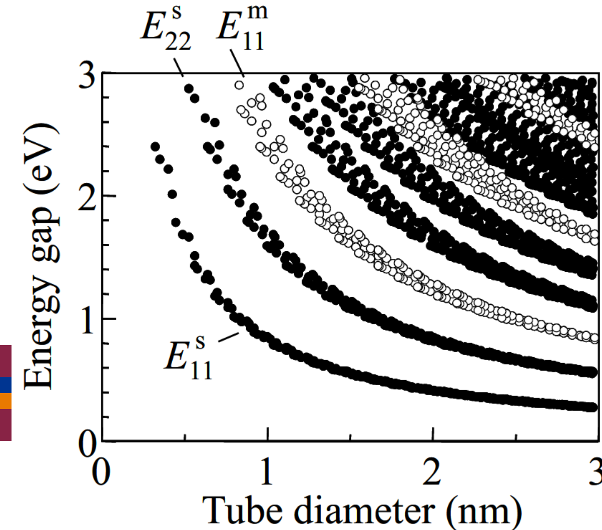
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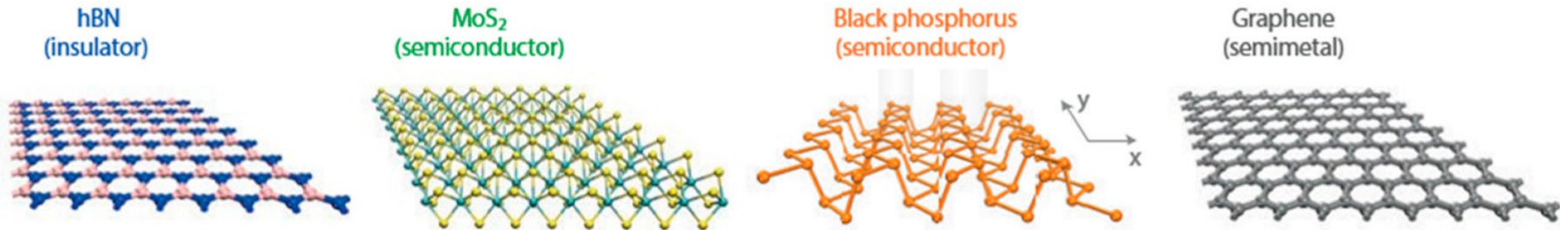
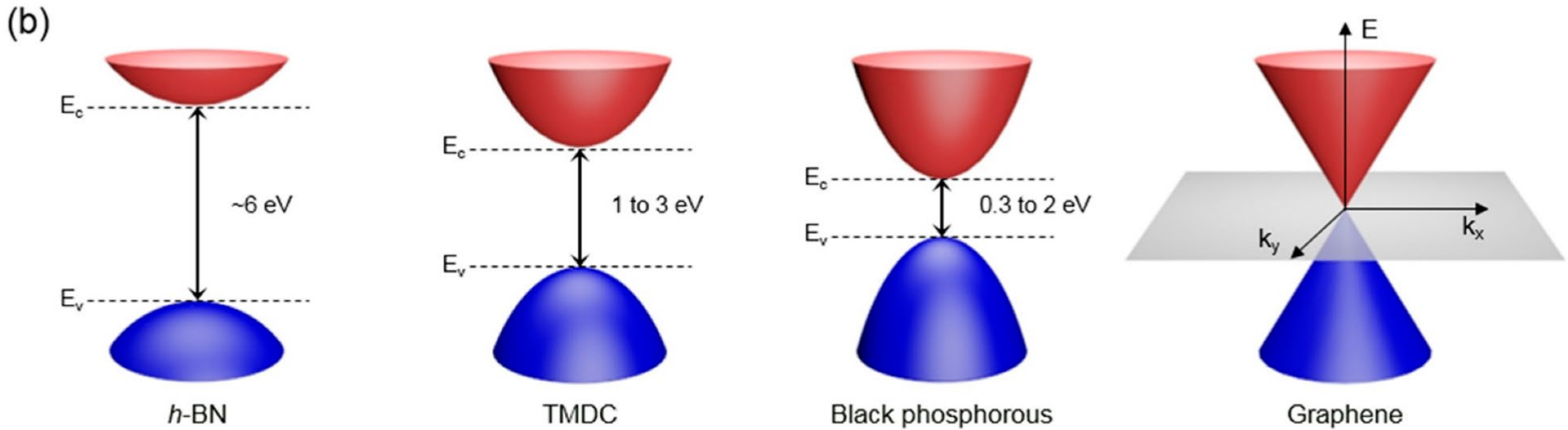
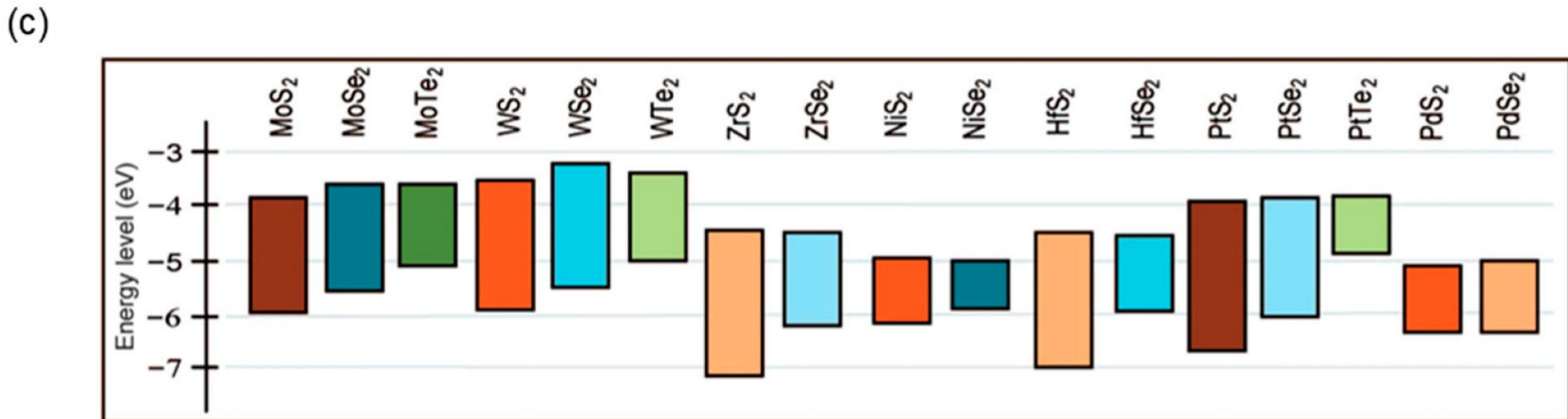
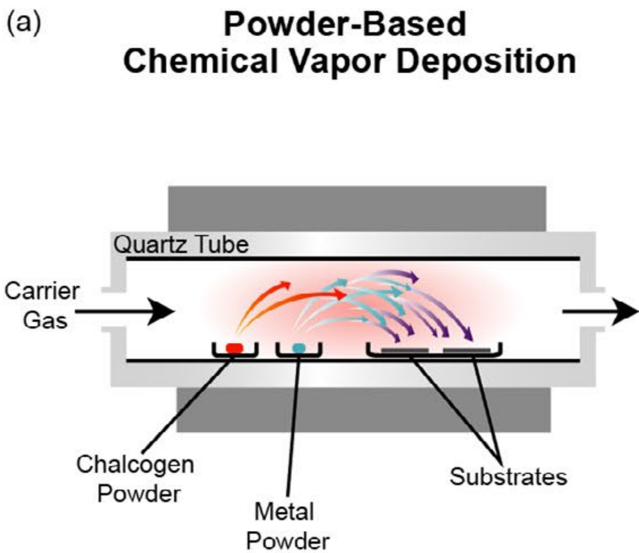
# Two-dimensional semiconductors

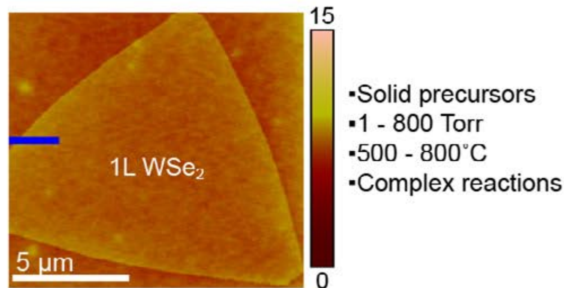


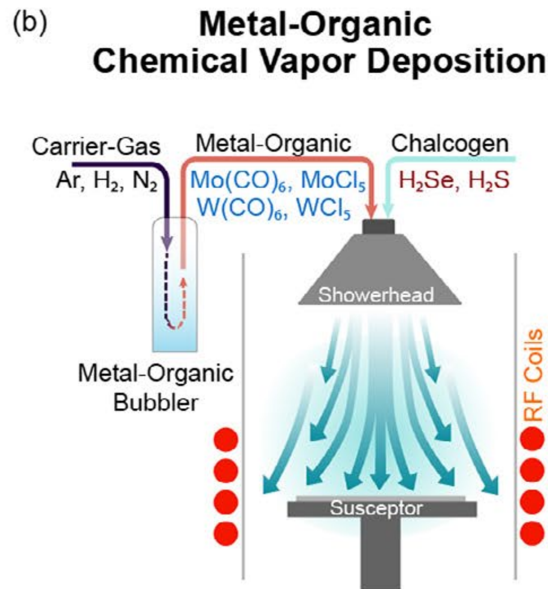



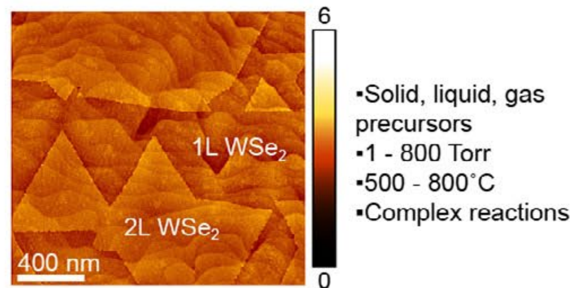
# Two-dimensional semiconductors: $WSe_2$, $MoSe_2$



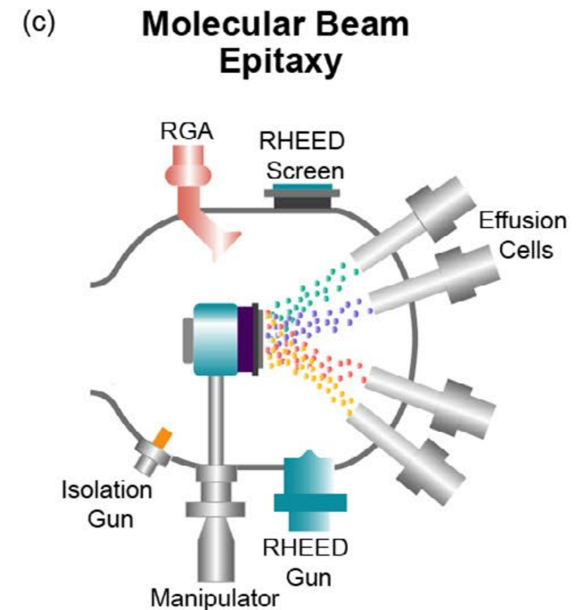



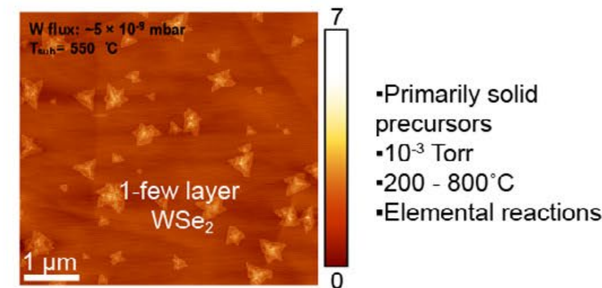

- Solid precursors
- 1 - 800 Torr
- 500 - 800°C
- Complex reactions




- Solid, liquid, gas precursors
- 1 - 800 Torr
- 500 - 800°C
- Complex reactions

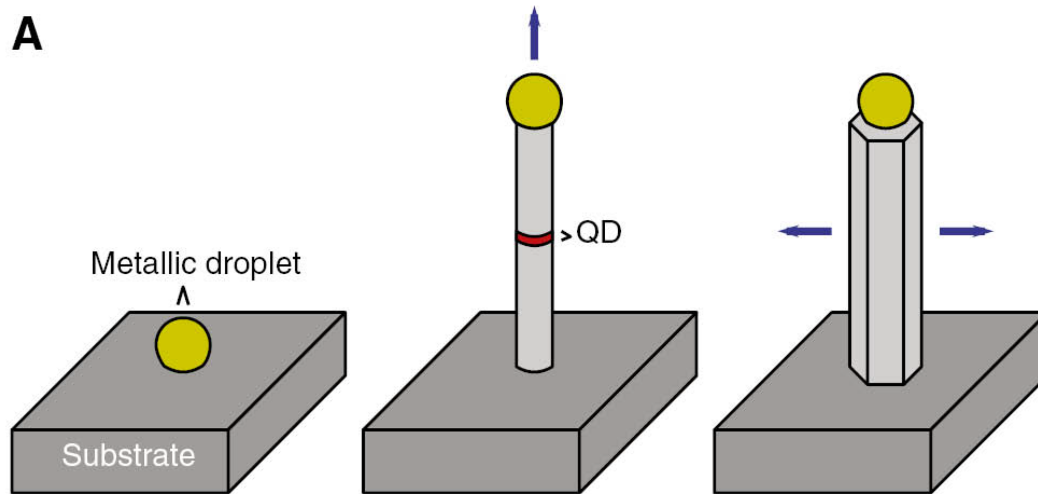


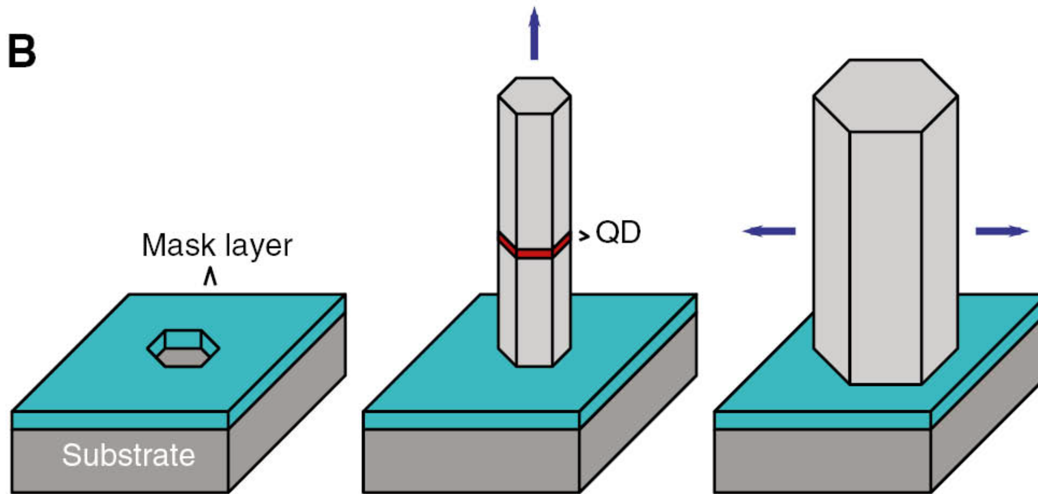

- Primarily solid precursors
- $10^{-3}$ Torr
- 200 - 800°C
- Elemental reactions

N. Briggs, 2D Materials **6**, 022001 (2019)

# Growth of vertical quantum wires




Vapor-liquid-solid




Selective area epitaxy

Mäntynen, Nanophotonics **8**, 747 (2019)



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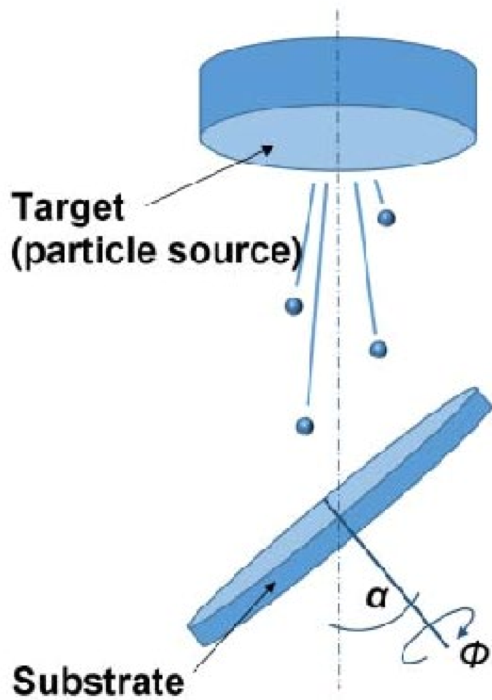
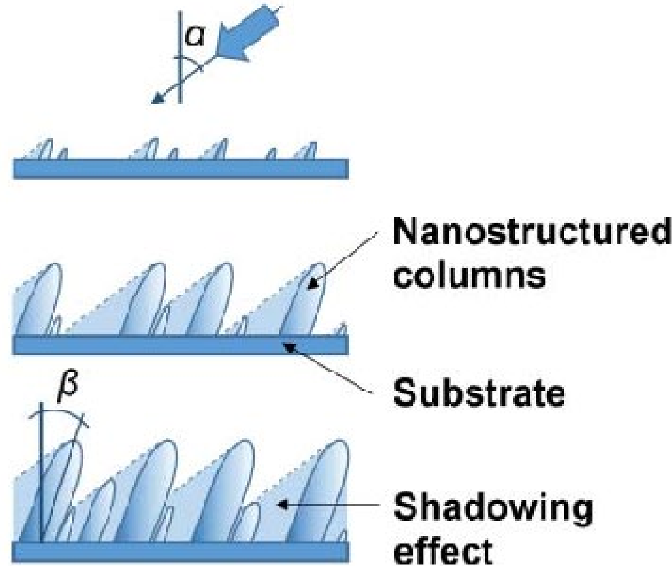
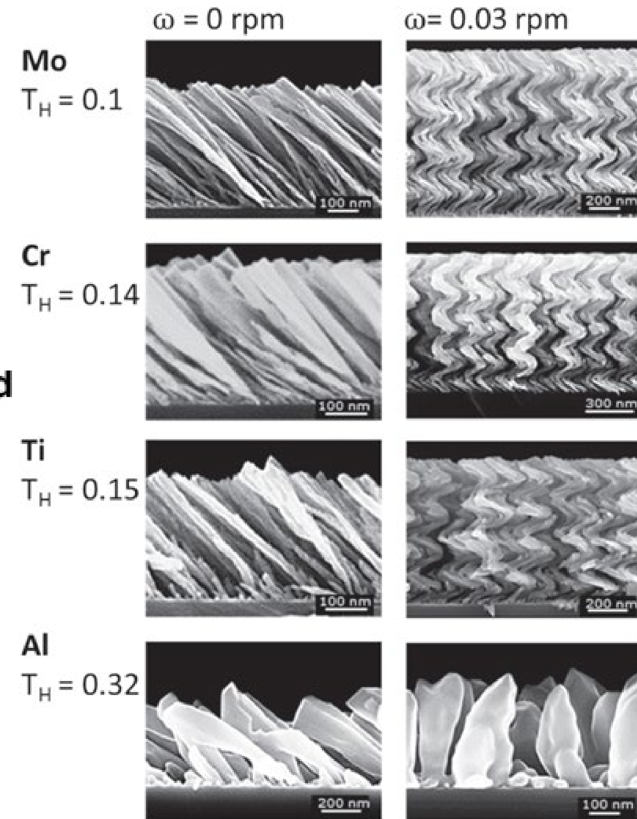
# Glancing angle deposition (GLAD) of quantum wires




Initial imperfections on the substrate
Shadow effect leads to tilted column growth.



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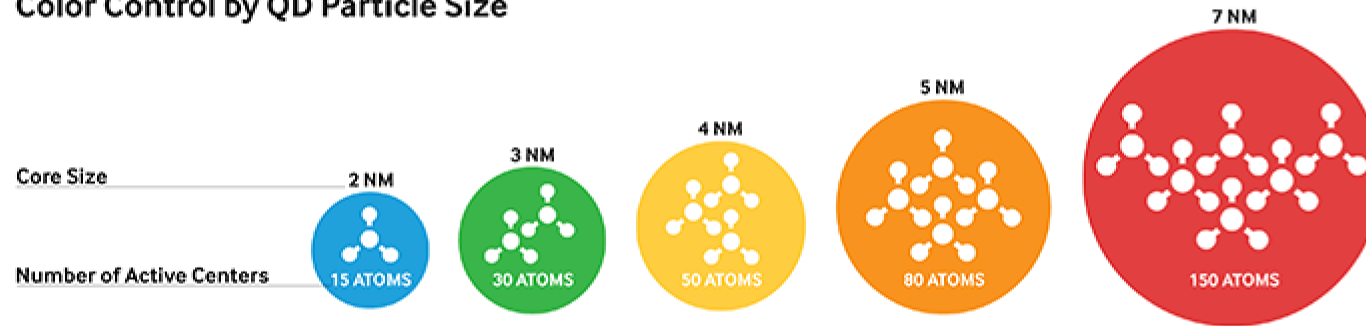
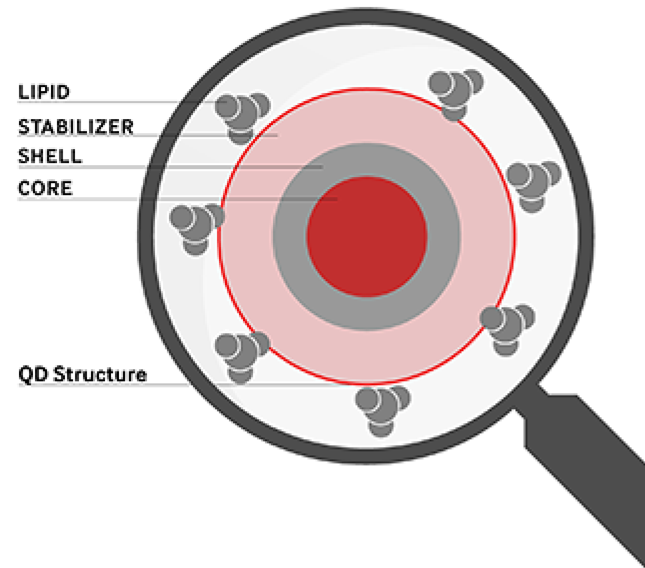
# Colloidal quantum dots





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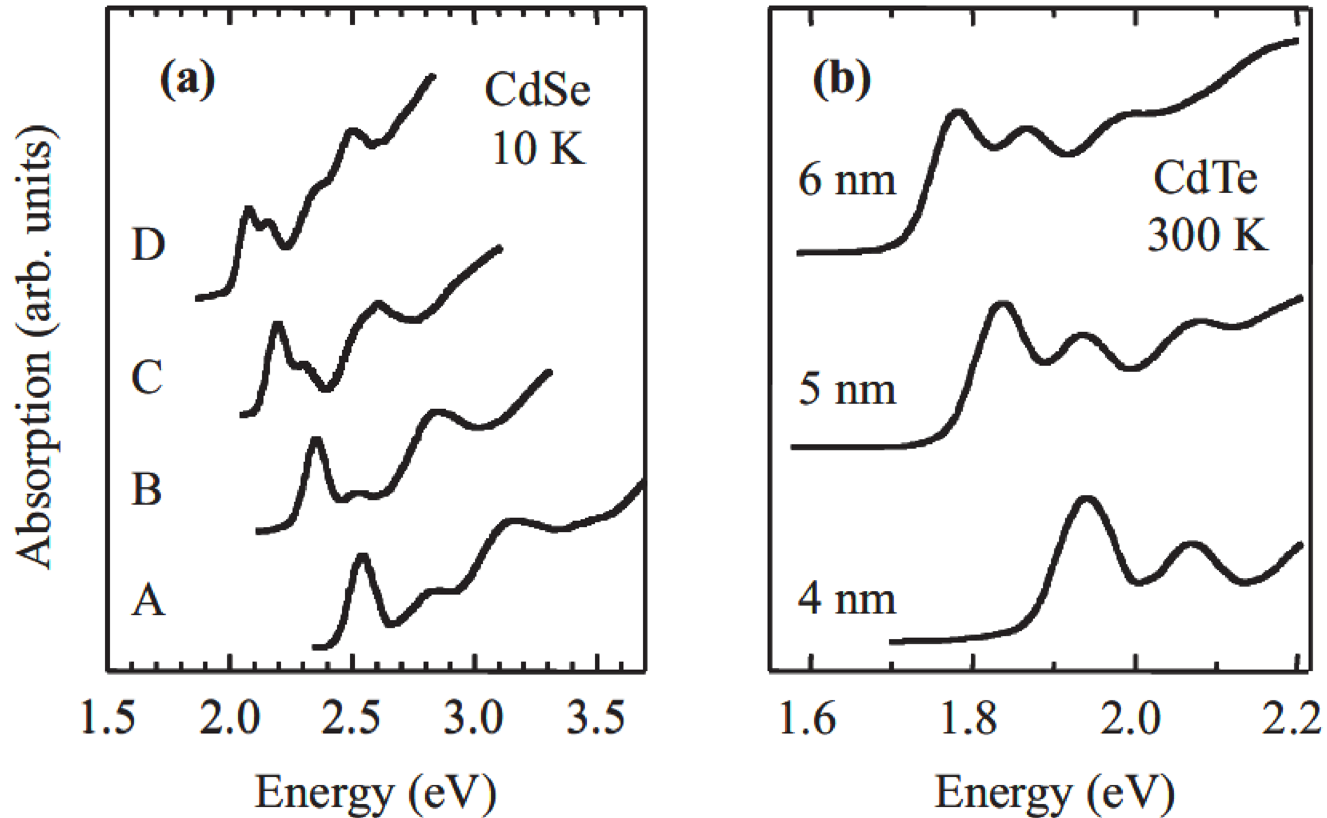
# Colloidal quantum dots




Fox, Chapter 6
C.R. Kagan, C.B. Murray, M.G. Bawendi, PRB **54**, 8633 (1996)



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# Epitaxial quantum dots

Stranski-Krastanov growth of semiconductor islands by MBE

Photoresist

[001]

[110]

[$\bar{1}$10]

(a) (001) GaAs substrate

(b) (111)A

InAs quantum dot

200 nm

GaAs

10 nm

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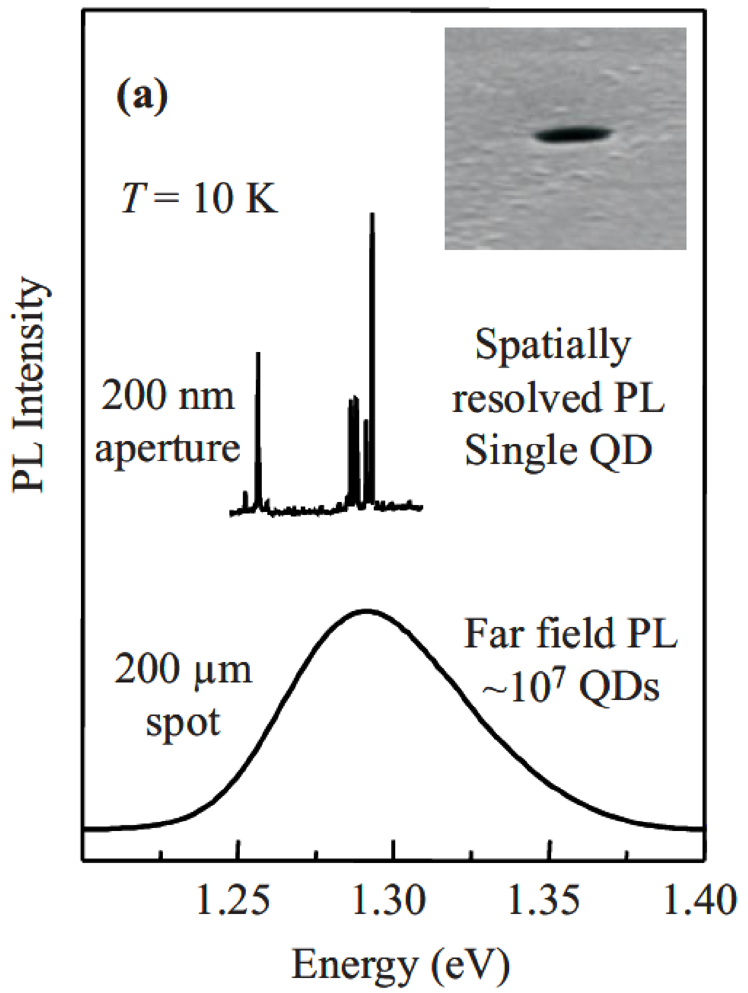
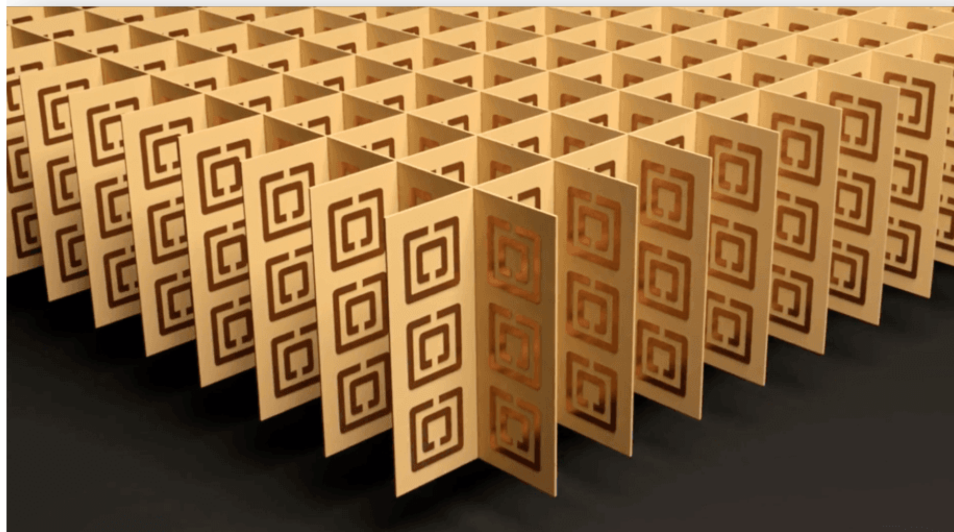
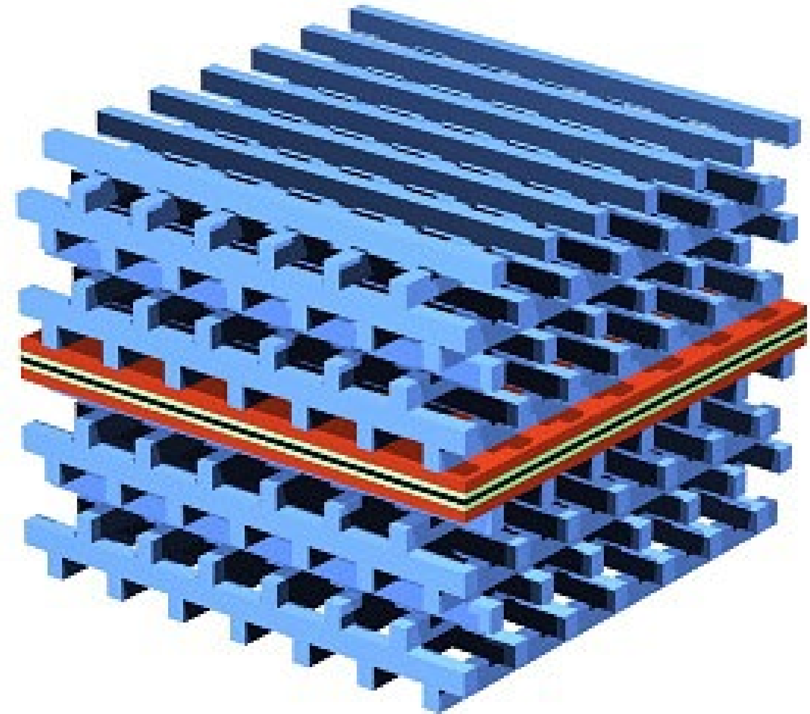


# Metamaterials



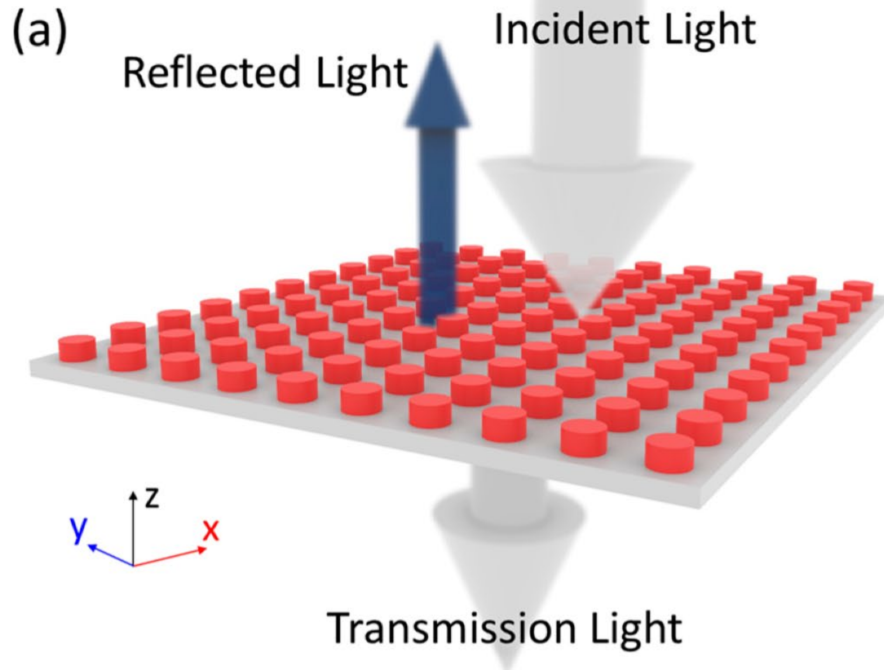


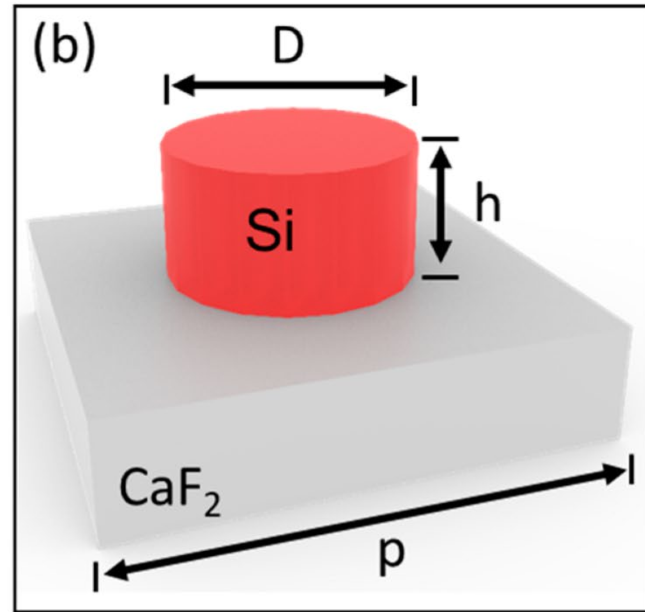
Faraday: $\vec{k} \times \vec{E}_0 = \omega\mu\mu_0\vec{H}_0$

Metamaterials are artificially structured three-dimensional materials with feature size less than a wavelength. They have unusual properties:
**Photonic band gaps, left-handed materials, negative index.**

Sajeev John
Shelby, APL **78**, 489 (2001)

# Metasurfaces




Metasurfaces can be designed to have reflection, absorption, and emission properties not possible in homogeneous materials. Especially useful for sensors and antennas.



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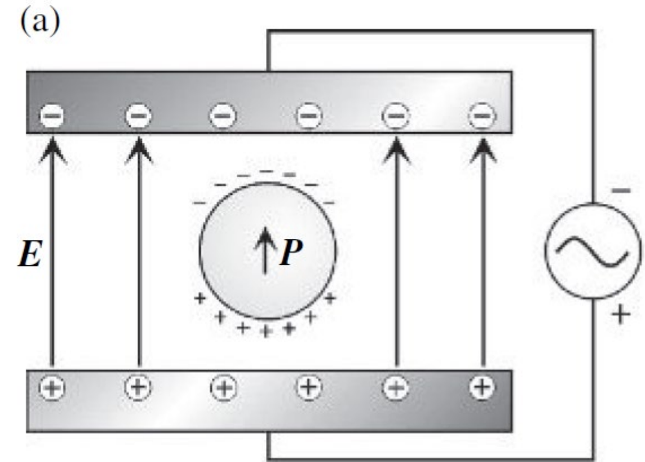
# Modeling Metamaterials and Metasurfaces

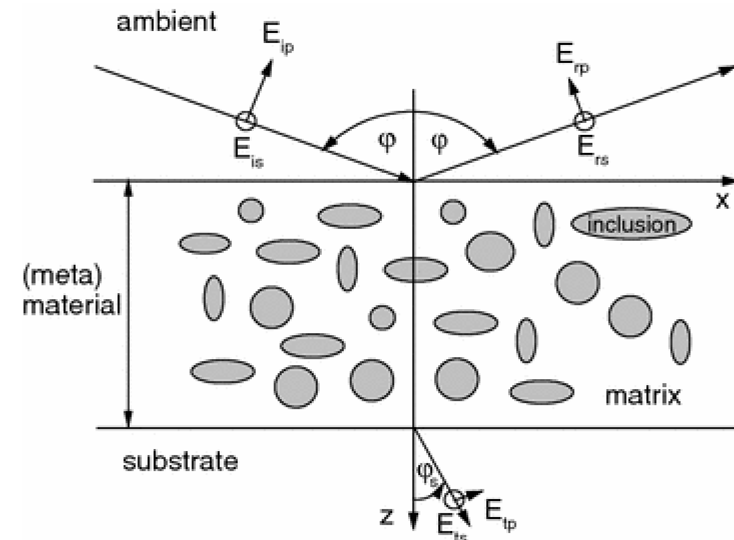
Bruggeman effective medium approximation (BEMA)

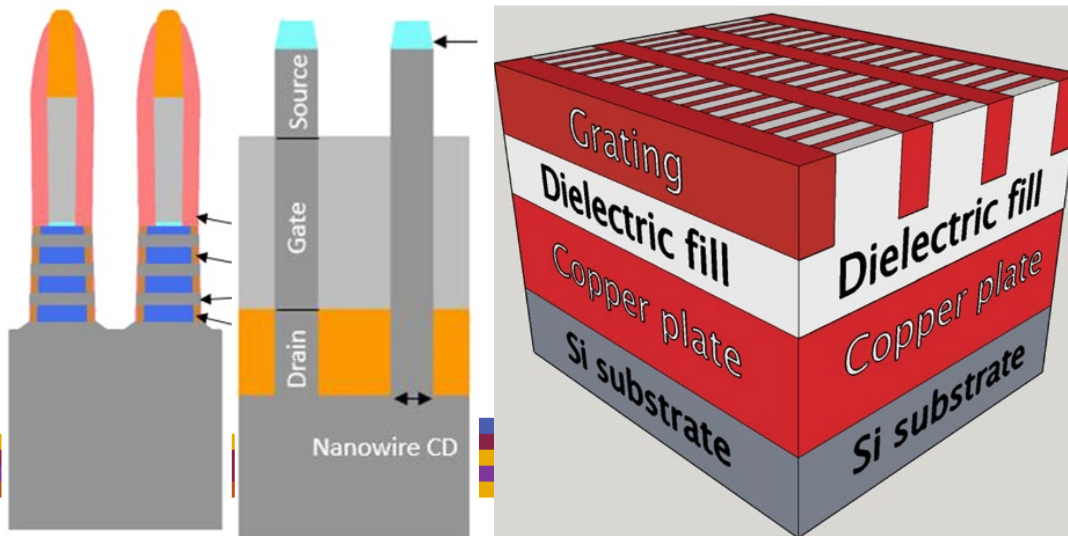
$$\sum_{i=1}^{n} f_i \frac{\varepsilon_i - \varepsilon}{\varepsilon_i + 2\varepsilon} = 0$$




Rigorous coupled wave approximation:

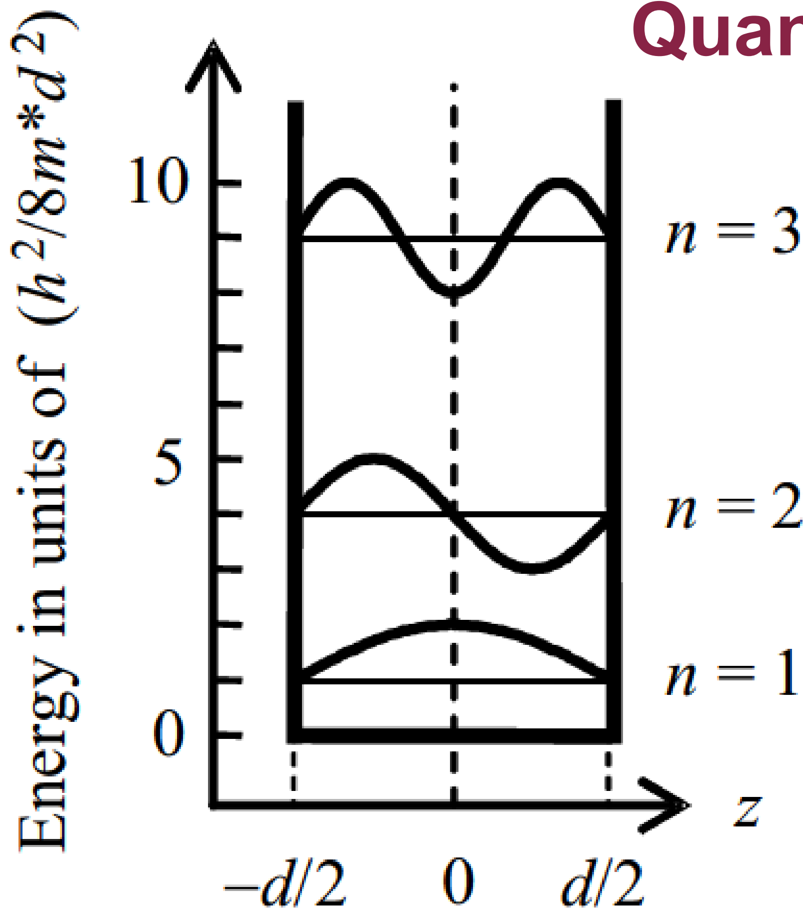
- Fourier method for periodic structures
- Numerical solution of Maxwell's equations




Schmidt, JAP **114**083519 (2013)
Diebold, APL Mater. **6**, 058201 (2018)

# Quantum well with infinite barriers




$$\Psi(x,y,z) = \psi(x,y)\varphi(z)$$

$$E\left(\vec{k},n\right) = \frac{\hbar^2 k^2}{2m} + \frac{\hbar^2}{2m}\left(\frac{n\pi}{d}\right)^2$$

Use carrier effective mass.

(Electrons more affected than holes).

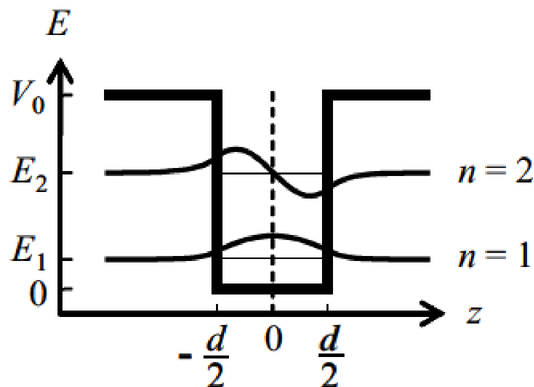
**Finite barrier:**

Confinement energies are lower.

Wave function leaks into barrier.

Finite number of bound states.

Solve numerically.

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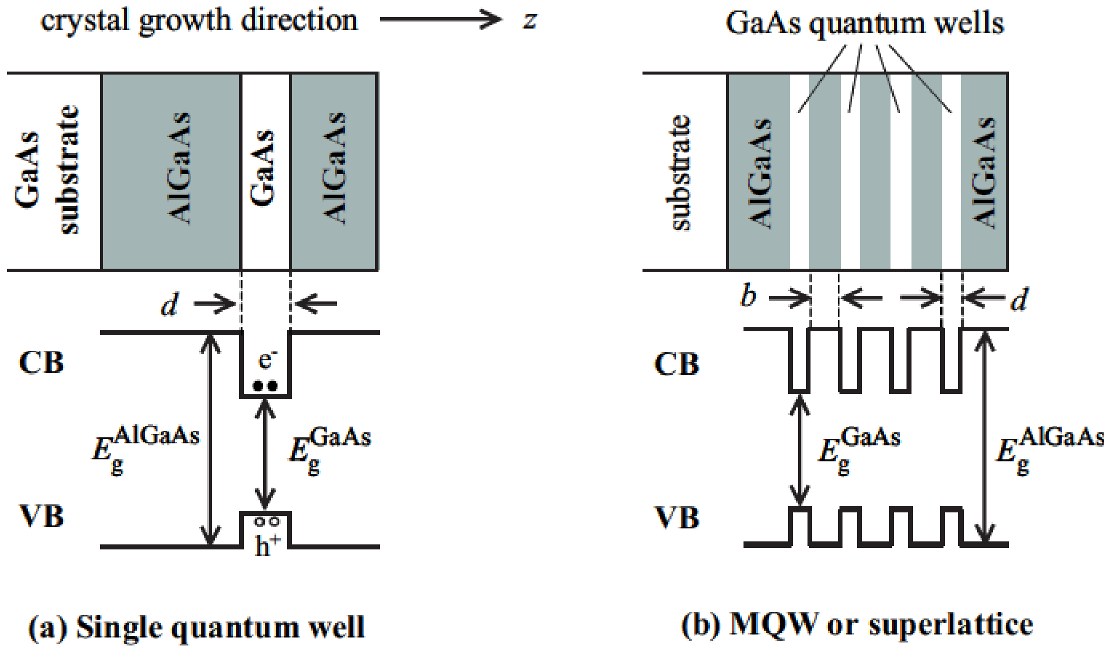
# Quantum wells and superlattices (or MQW)




(a) Single quantum well

(b) MQW or superlattice

Quantum well: wave function entirely contained in one well.

Superlattice: wave function leaks through the barrier into the next well (barrier thin or low).

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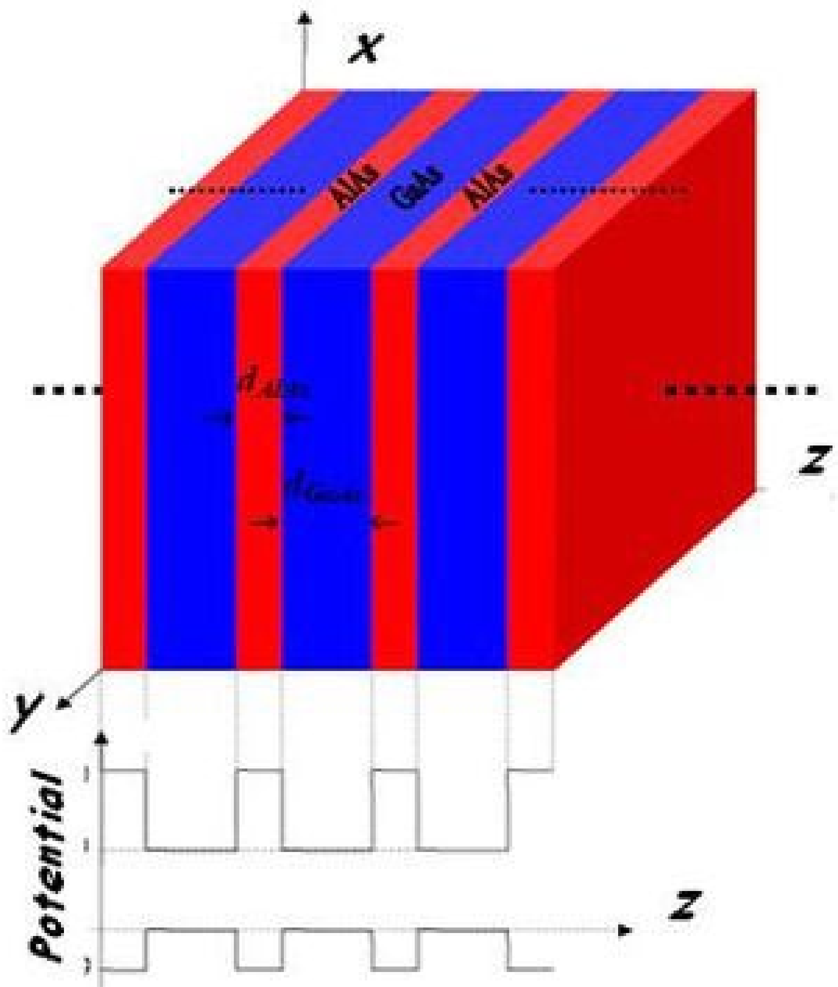
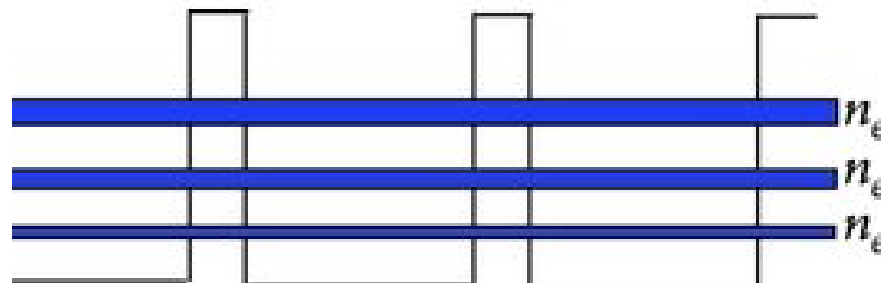
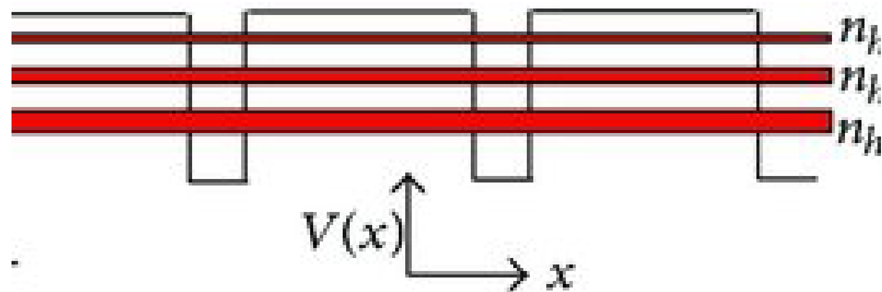
# Superlattice minibands




CB






VB



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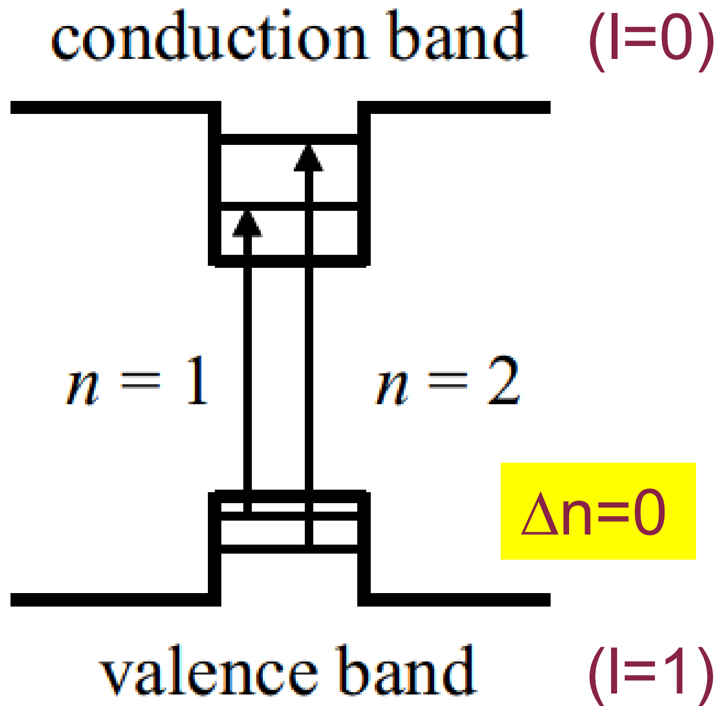
# Enhanced absorption/emission in quantum structures

$$\frac{1}{\tau} = \frac{2\pi}{\hbar} |\langle f | H_{eR} | i \rangle|^2 g_{fi}(\hbar\omega)$$

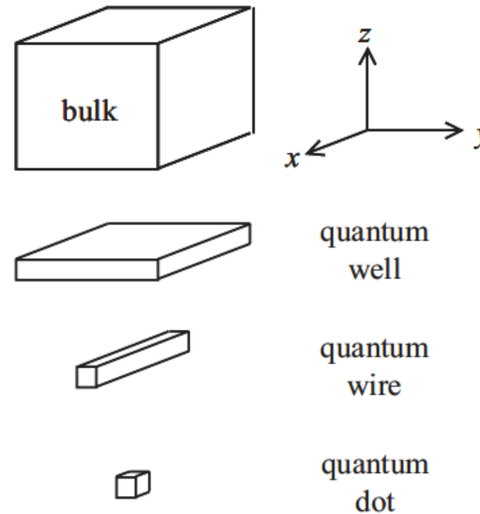
Confinement enhances overlap matrix element

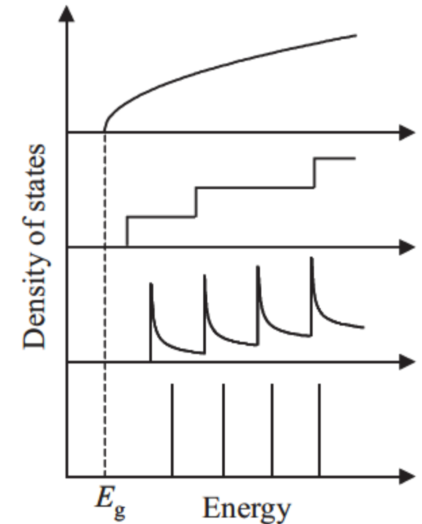



(l=0)

Δn=0

(l=1)




Modified density of states

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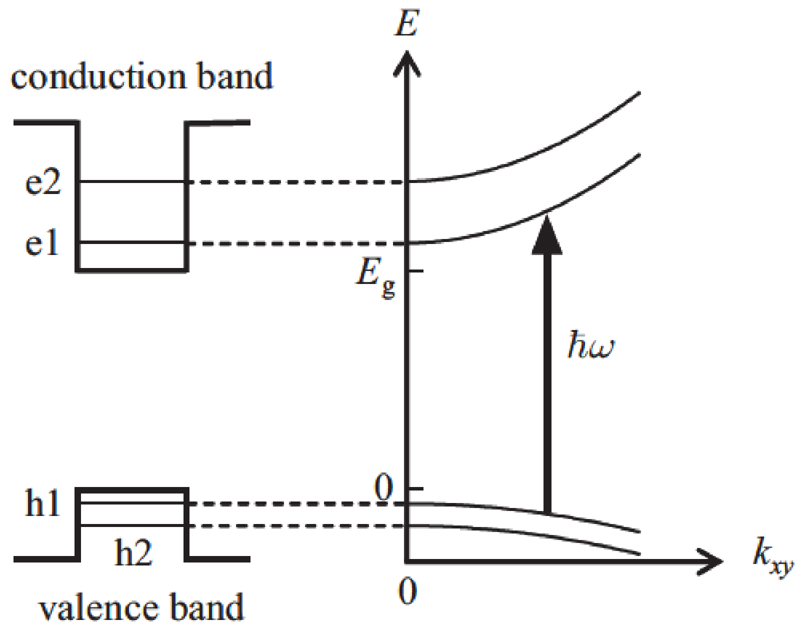
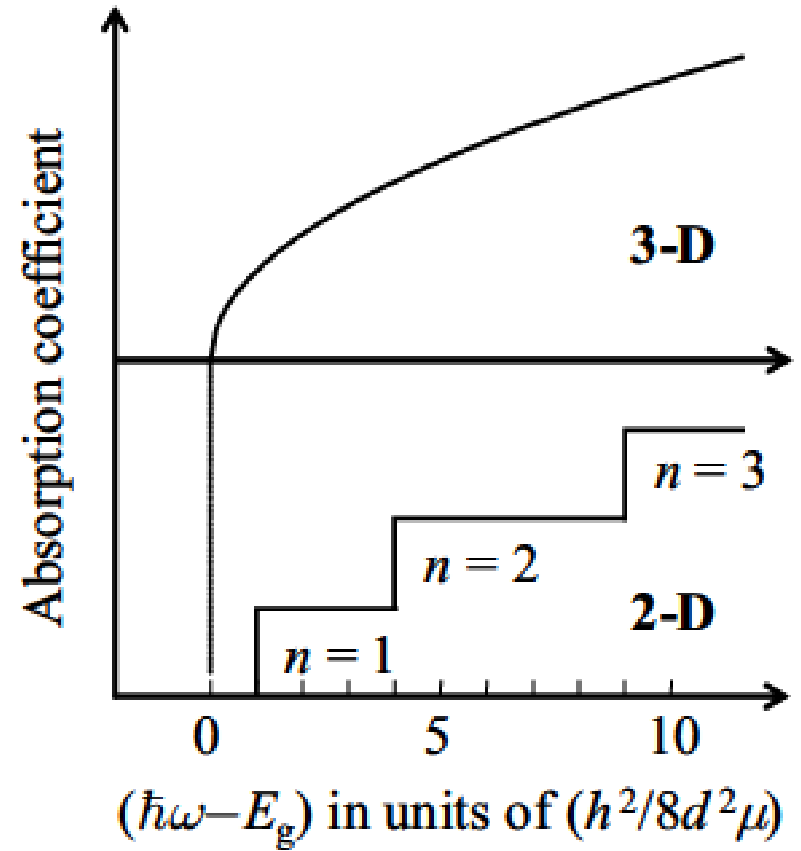
# Quantum well absorption




Electron and hole subbands

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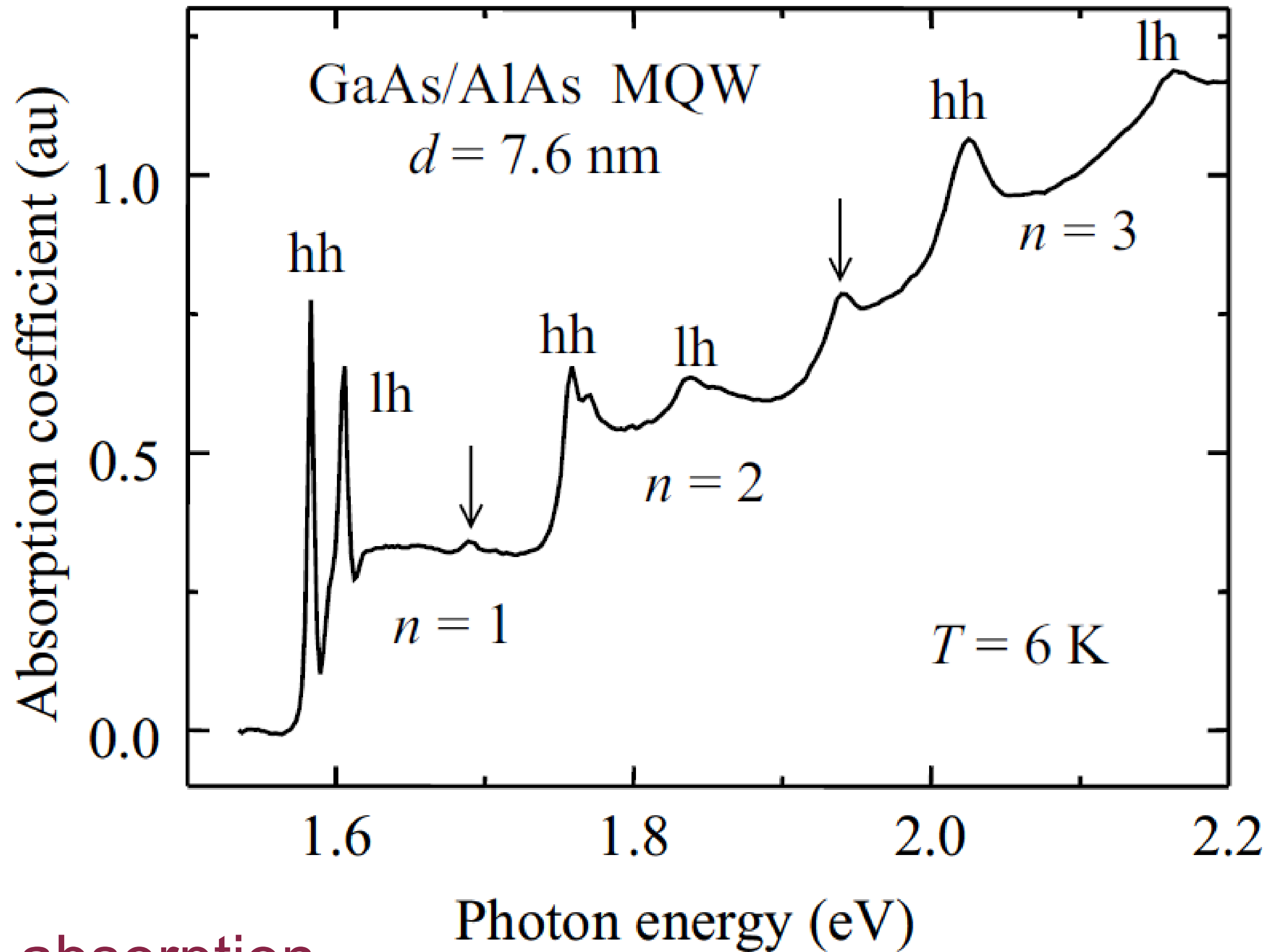
# Quantum well absorption in GaAs/AlAs MQW




Steps: 2-D absorption

Peaks: Discrete excitons (arrows show forbidden transitions)

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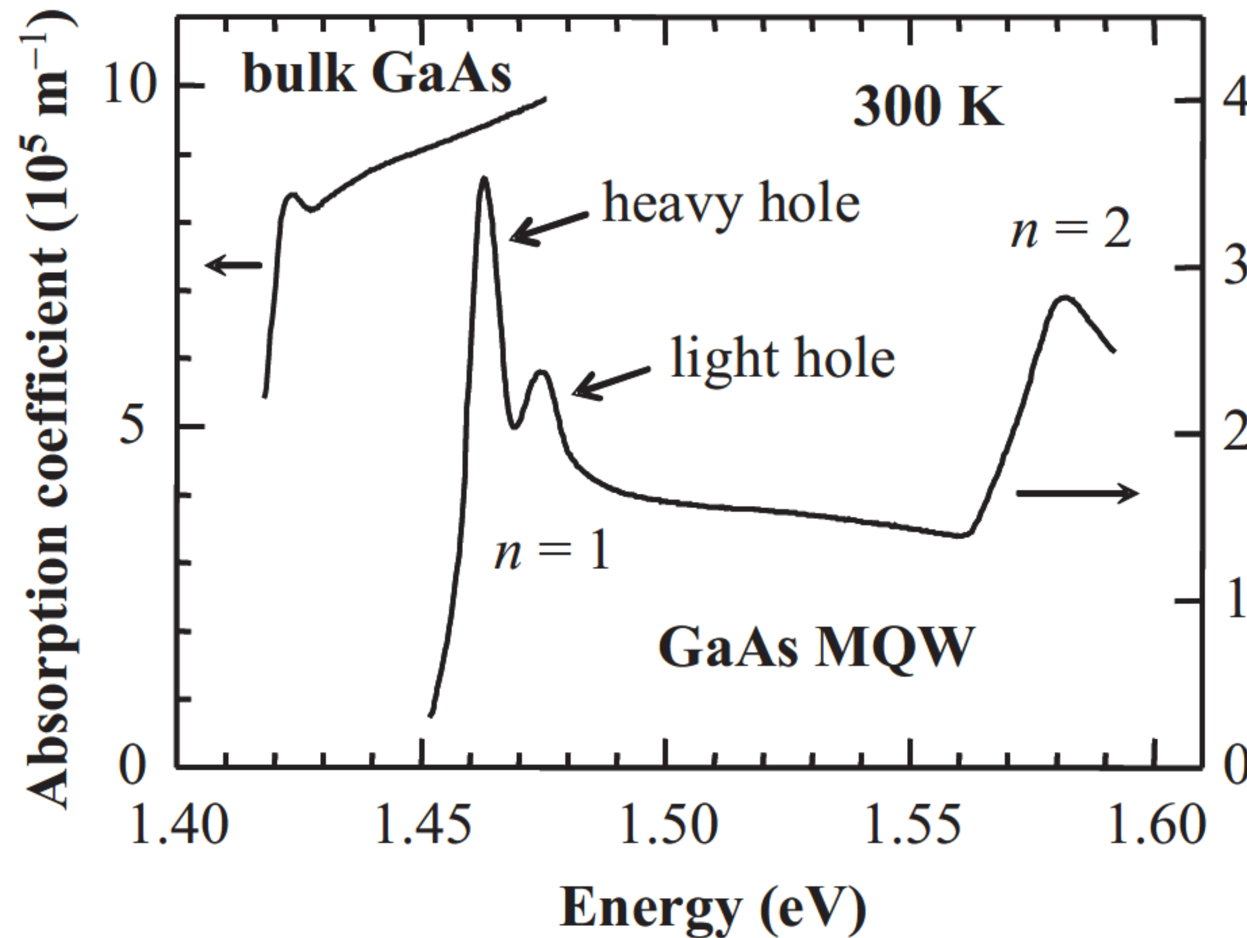
# Excitonic effects are enhanced in quantum wells




Bulk GaAs:
$R_X$=4.2 meV

Here (d=10 nm):
$R_X$=10 meV
Increased overlap integral

hh/lh splitting:
Confinement or strain ???

$$E\left(\vec{k},n\right)=\frac{\hbar^2 k^2}{2m}+\frac{\hbar^2}{2m}\left(\frac{n\pi}{d}\right)^2$$

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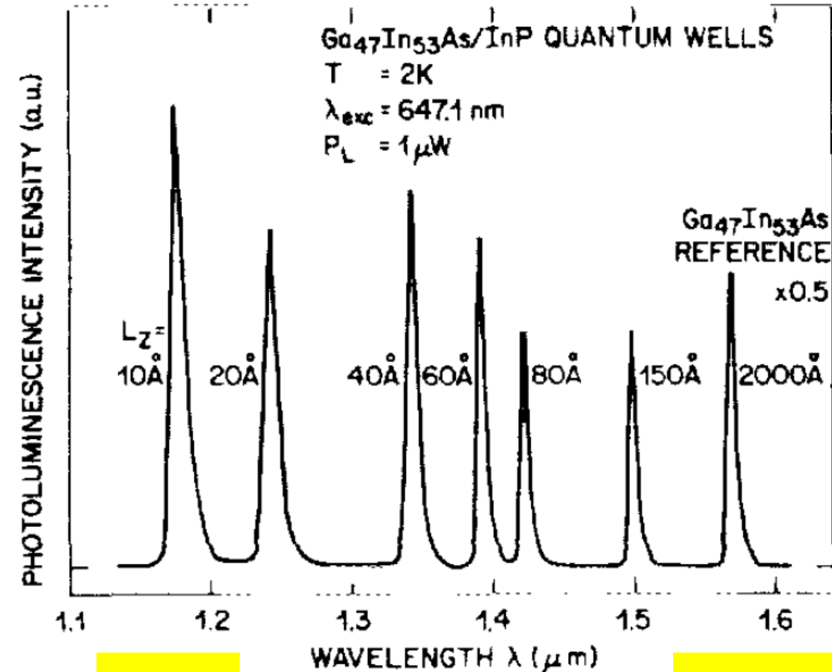
# Confinement shifts in quantum wells

$$\hbar\omega = E_g + E_{hh1} + E_{e1}$$




thin

thick

$$\Delta E = \frac{\hbar^2}{2m}\left(\frac{n\pi}{d}\right)^2$$

INFINITE WELL APPROXIMATION

$\Delta E_C/\Delta E_V$ = 80/20

50/50

20/80

Ga47In53As/InP QUANTUM WELL T = 2K

● SAMPLE A

▲ SAMPLE B

■ SAMPLE C

▼ SAMPLE D

× SAMPLE E

PHOTOLUMINESCENCE ENERGY SHIFT $\Delta E_{1n}$ (meV)

THICKNESS OF QUANTUM WELL $L_Z$ (Å)

W.T. Tsang & E.F. Schubert,
Appl. Phys. Lett. **49**, 220 (1986)



STATE

# Intersubband transitions, quantum cascade lasers

$n = 2$

electrons

$\hbar\omega$

$n = 1$

$\Delta$n odd

Energy

INJECTION

3

2

1

1 period

Infrared detectors and lasers

Light polarized along z (beam in quantum well plane)

Fox, Chapter 6

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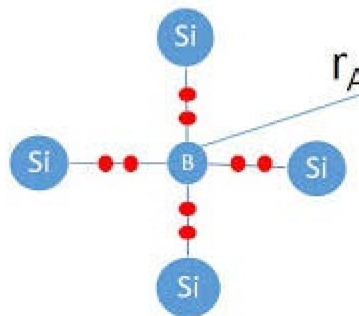
# Defects

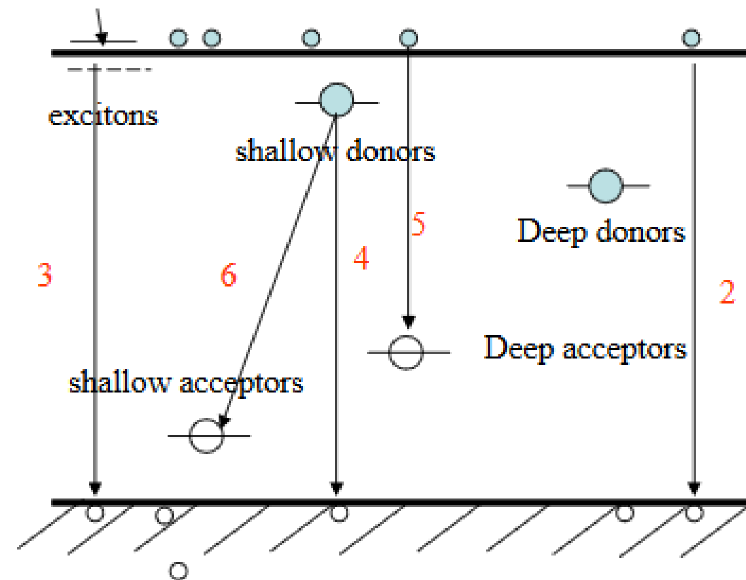
| | | |
|---|---|---|
| **Vacancy** | $V_A$ | missing atom |
| **Interstitial** | $I_A$ | extra atom between lattice sites |
| **Substitutional** | $C_A$ | defect atom replaces host atom |
| **Antisite** | | atoms are switched |
| Frenkel defect pair | $V_A$-$I_A$ | atom moved to interstitial site |

More complex defects (combinations)

**Donor**: Substitutional defect, adds an electron
**Acceptor**: Substitutional defect, adds a hole
**Isoelectronic**: Impurity has same number of electrons as host

**Shallow** and **deep** defects.

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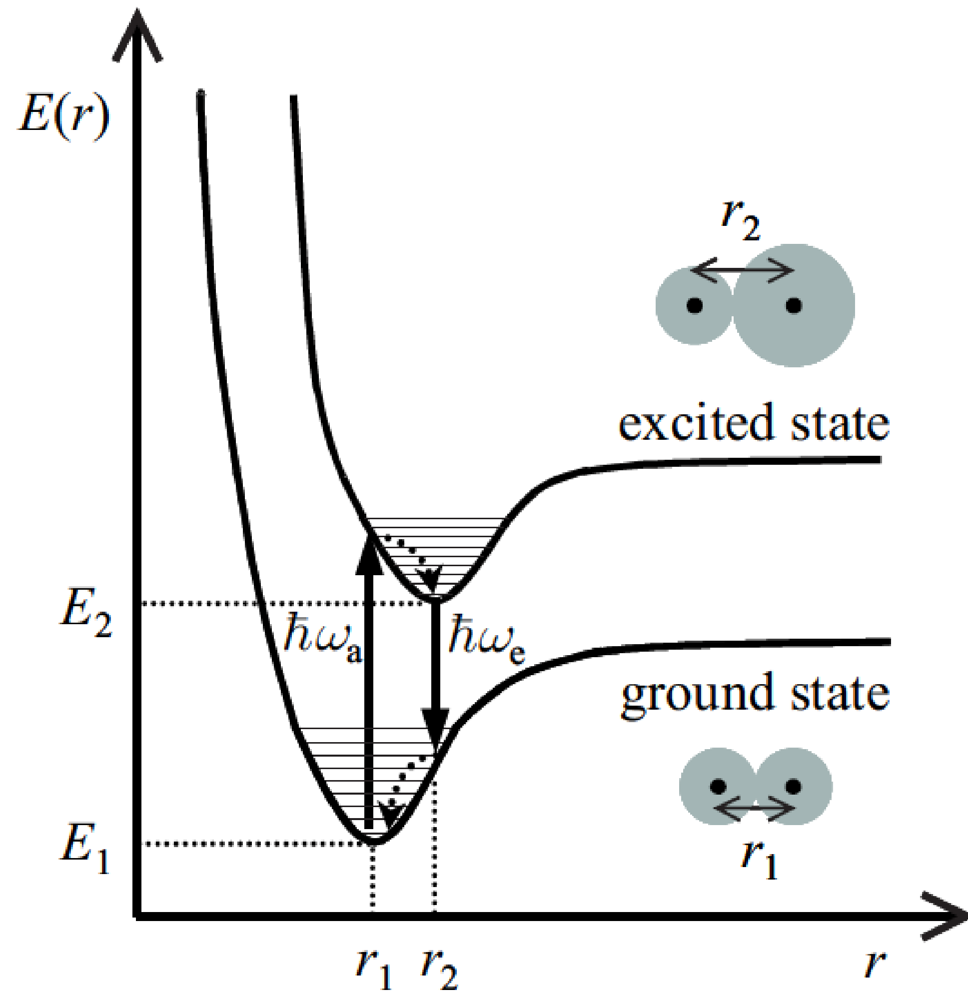
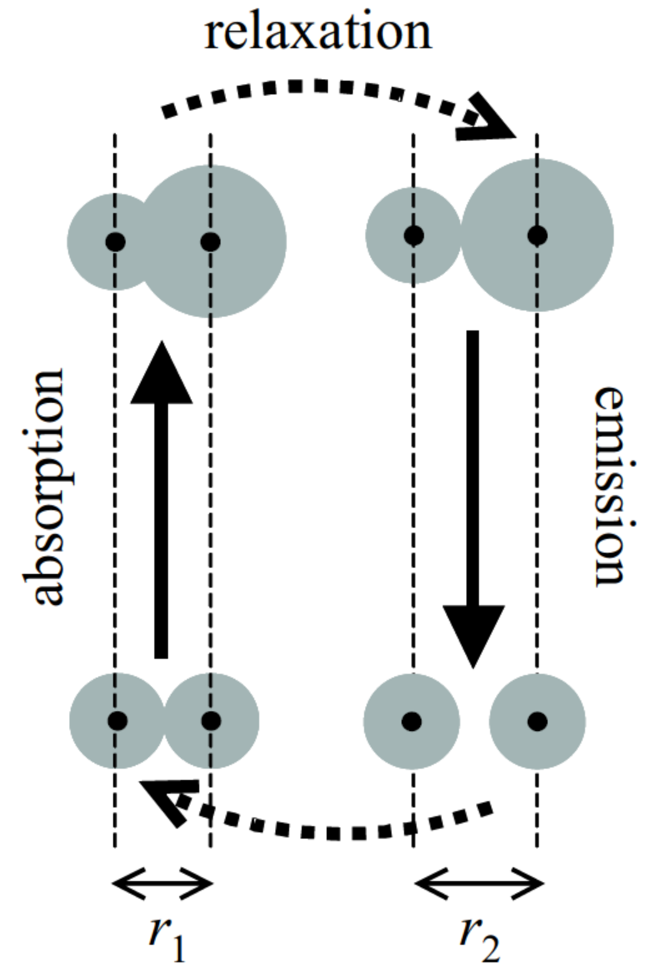
# Defects: Frank-Condon principle




Born-Oppenheimer approximation

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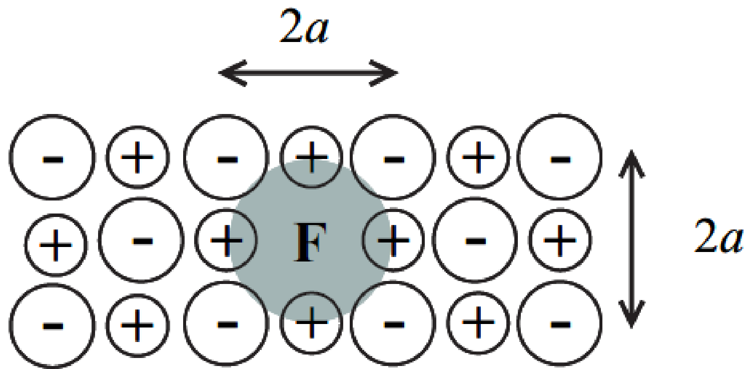
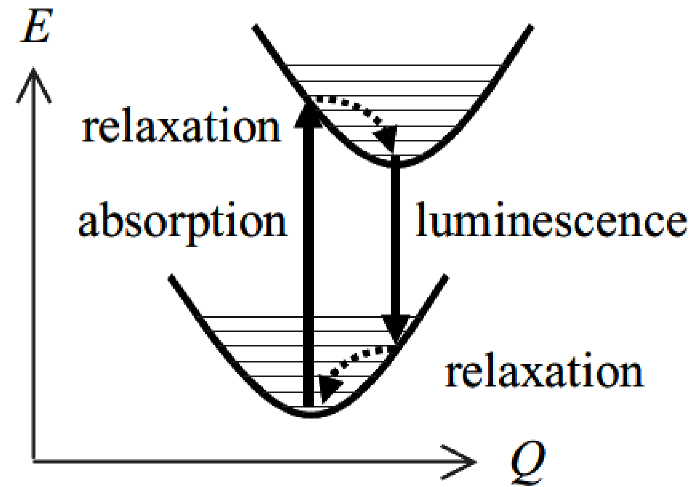
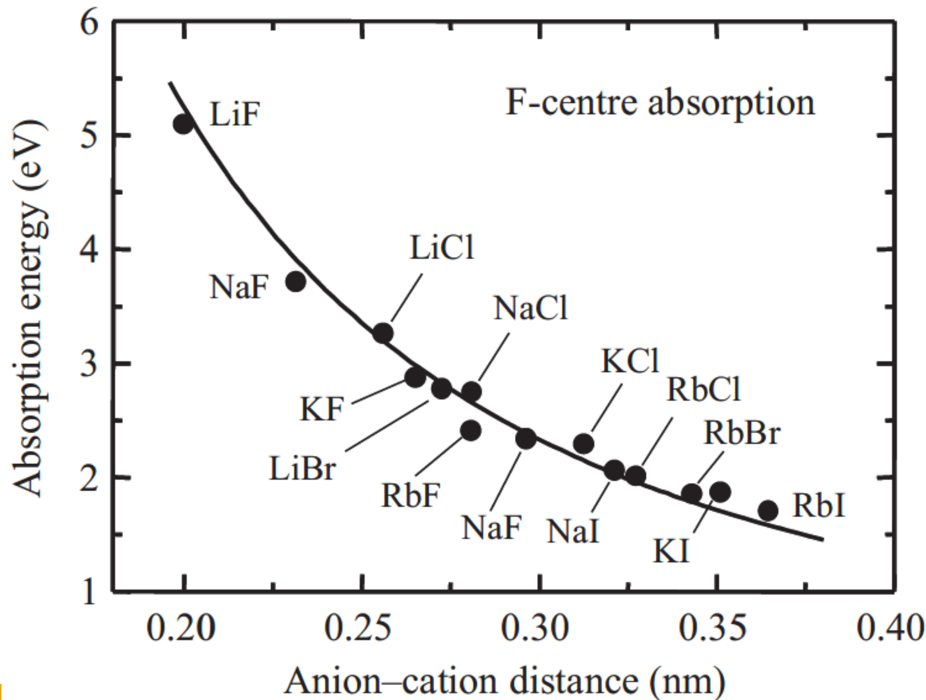
# Defects: Color centers (F-centers)




Alkali halide: large-gap insulator
Vacancy: positively charged hole

F-center: electron trapped at anion vacancy.

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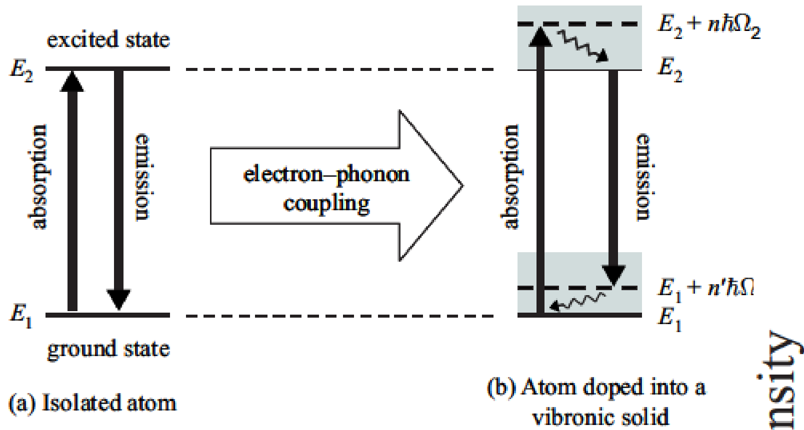
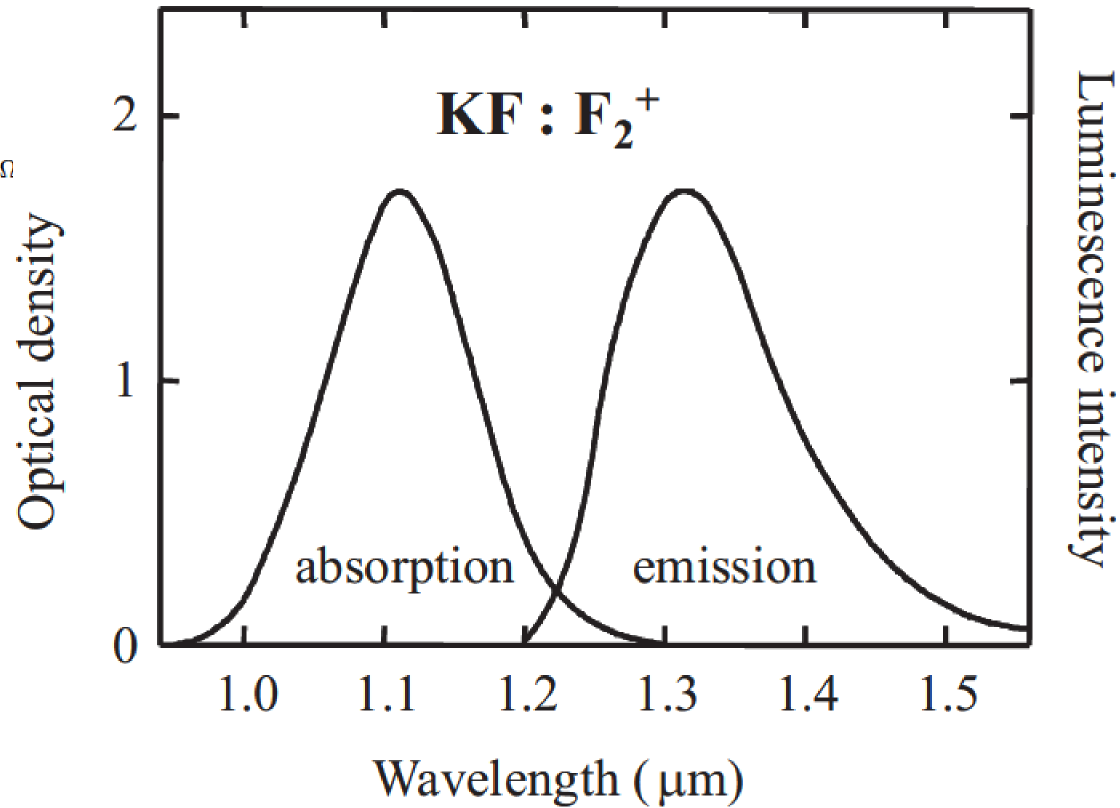
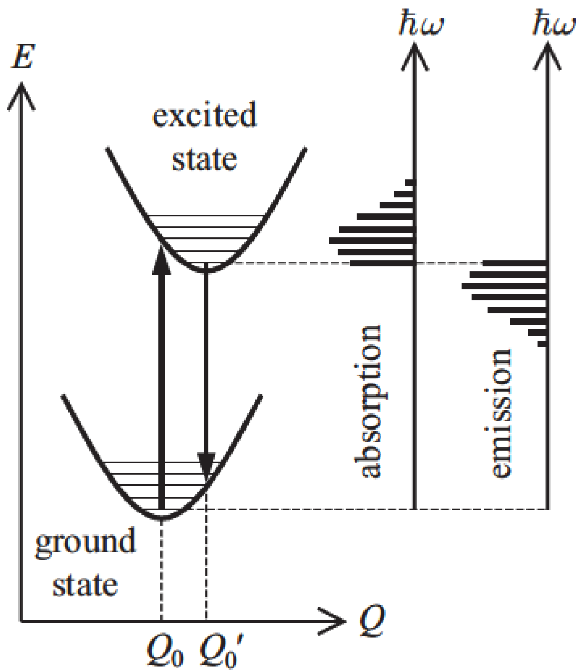
# Defects: Shift between absorption and emission




Two vacancies, one electron
Color-center lasers

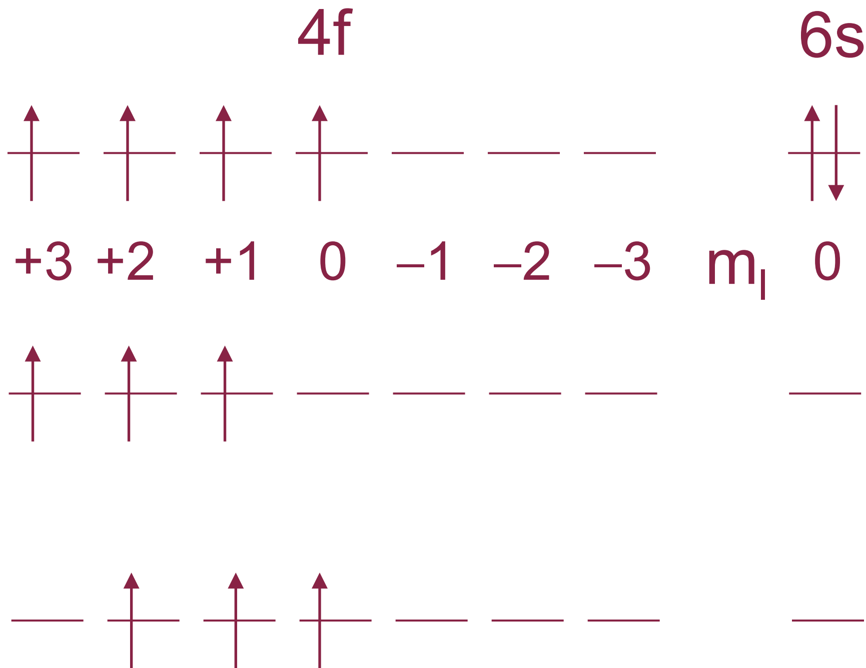
Fox, Chapter 8,9

# Hund's Rules (applied to Neodymium, Z=60)

1. Maximize S
2. Maximize L
3. If shell is less than half full, J=L–S
   If shell is more than half full, J=L+S.

$^{2S+1}L_J$

Russell-Saunders (LS) coupling




atom: $^5I_4$

Nd: $4f^4\ 6s^2$ S=2, L=6, J=4

3+ ion ground state: $^4I_{9/2}$

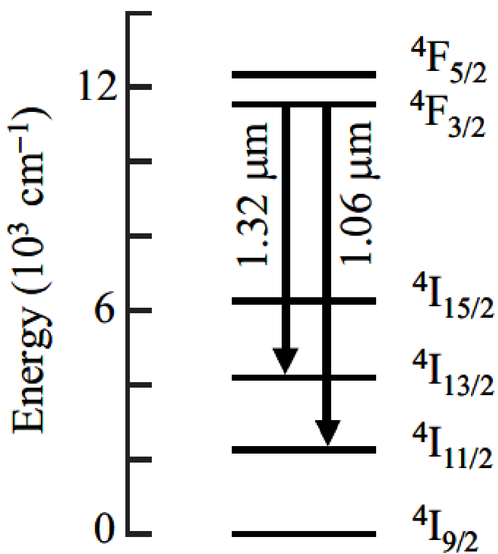
$Nd^{3+}$: $4f^3\ 6s^0$ S=3/2, L=6, J=9/2

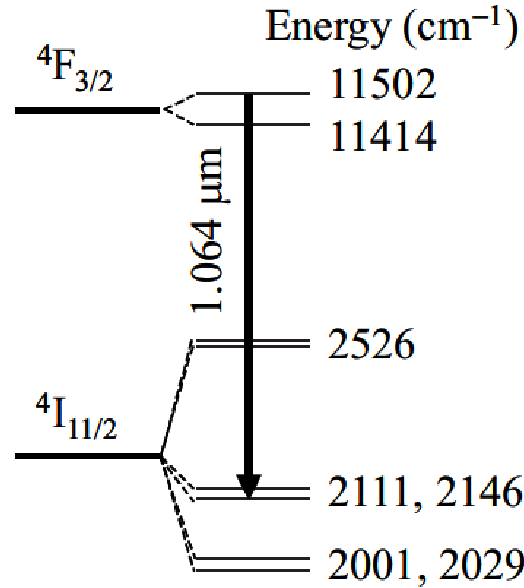
3+ ion excited state: $^4F_{3/2}$

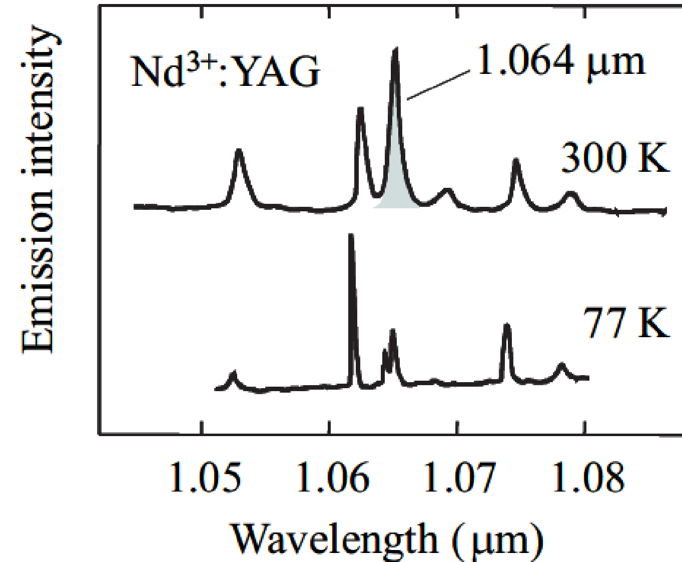
$Nd^{3+}$: $4f^3\ 6s^0$ S=3/2, **L=3**, J=3/2

Fox, Chapter 9

# Defects: Rare earth metal ions in insulator (Nd:YAG)




Atomic configuration

Weak crystal-field splitting

YAG: $Y_3Al_5O_{12}$

ΔJ=4 or 5 forbidden

Transition weakly allowed with crystal-field splitting

Nd:YAG lasers
YAG: 1064 nm
$YLiF_4$: 1053 nm
Broad (120 GHz)

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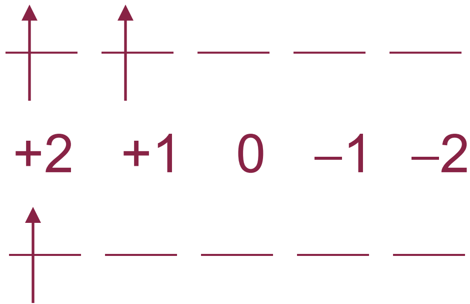
# Defects: Transition metal ions (Ti:sapphire)

3d 4s

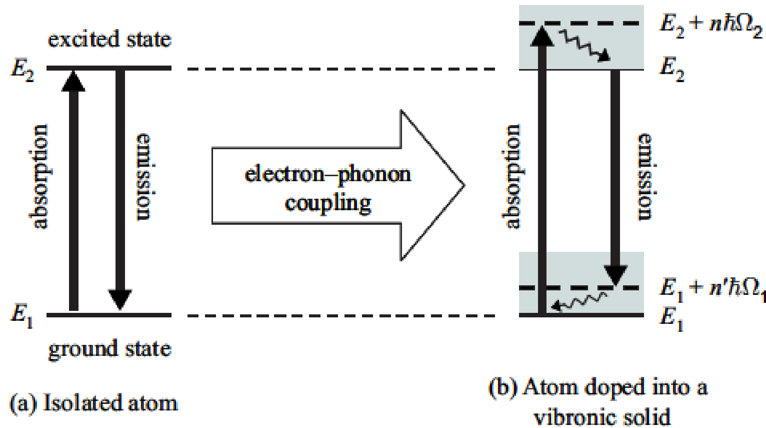



atom: $^3F_2$

Ti: $3d^2\ 4s^2$ S=1, L=3, J=2

ion: $^2D_{3/2}$

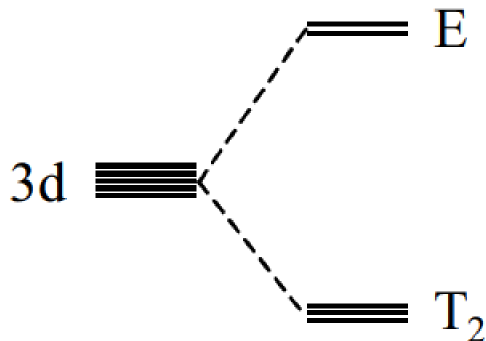
$Ti^{3+}$: $3d^1\ 4s^0$ S=1/2, L=2, J=3/2




(a) Isolated atom

(b) Atom doped into a vibronic solid

Strong vibronic coupling
Broad lines (gain spectrum)
Good for ultrafast lasers




Strong crystal-field splitting
Depends on crystal environment

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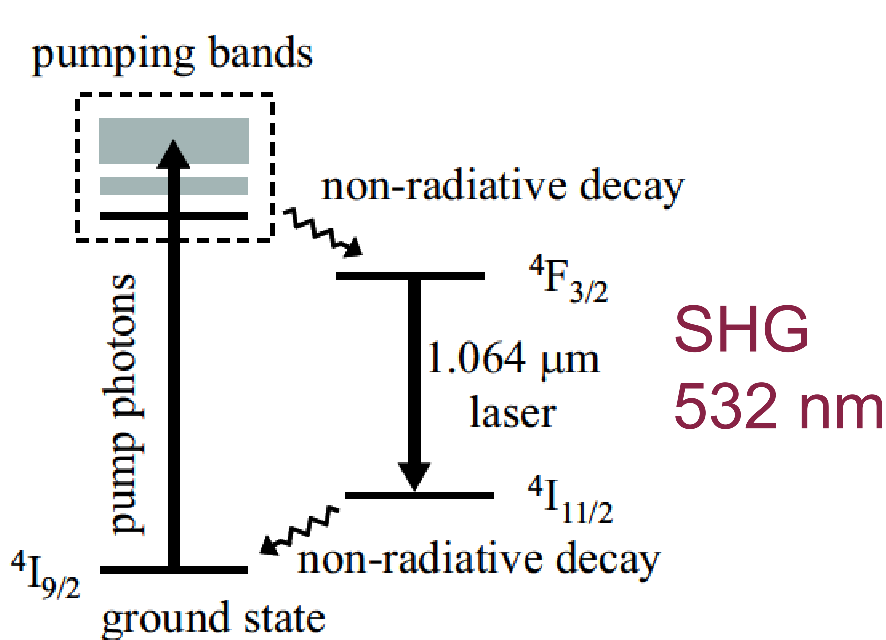
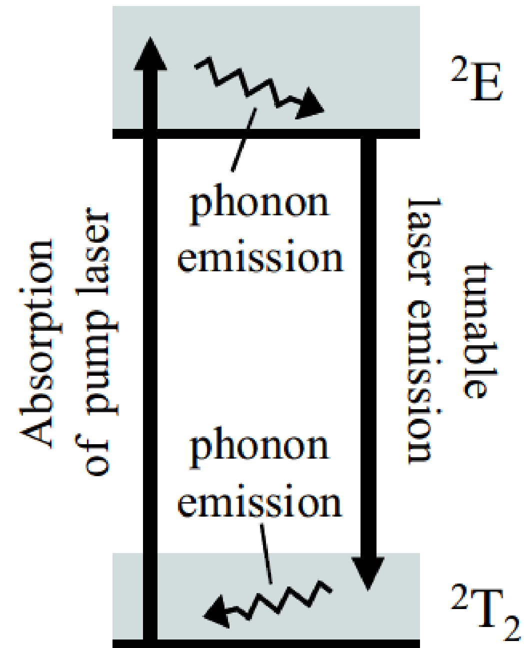
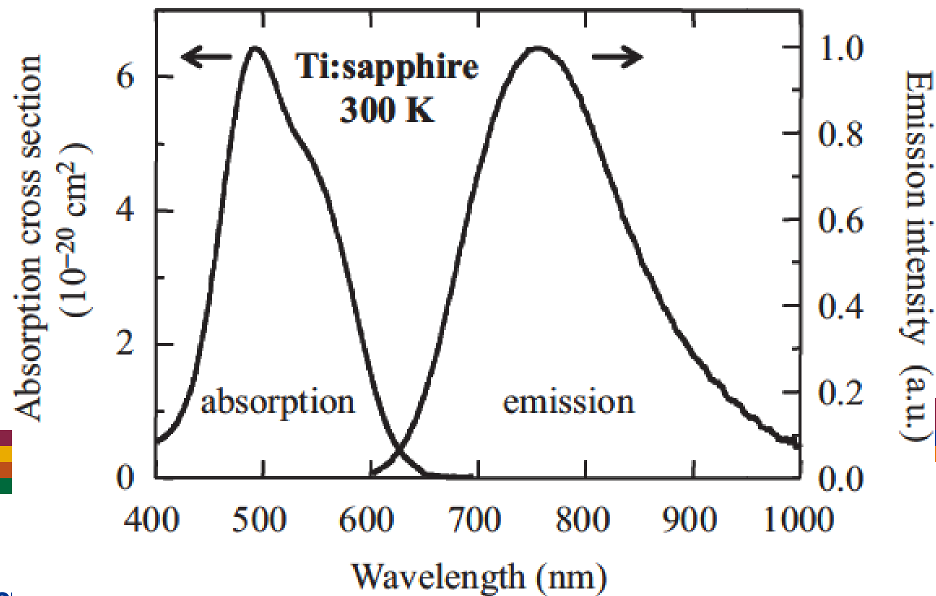
# Solid-State Lasers: Ti:sapphire and Nd:YAG




SHG
532 nm




Pumping with
- Flash lamp (pulsed)
- Xe lamp (CW)
- GaAs diode (efficient)

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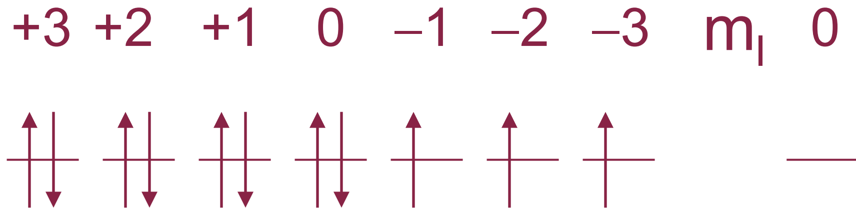
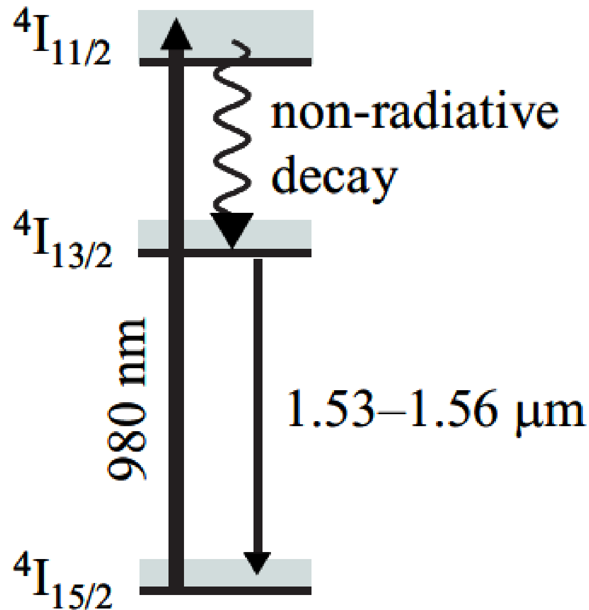
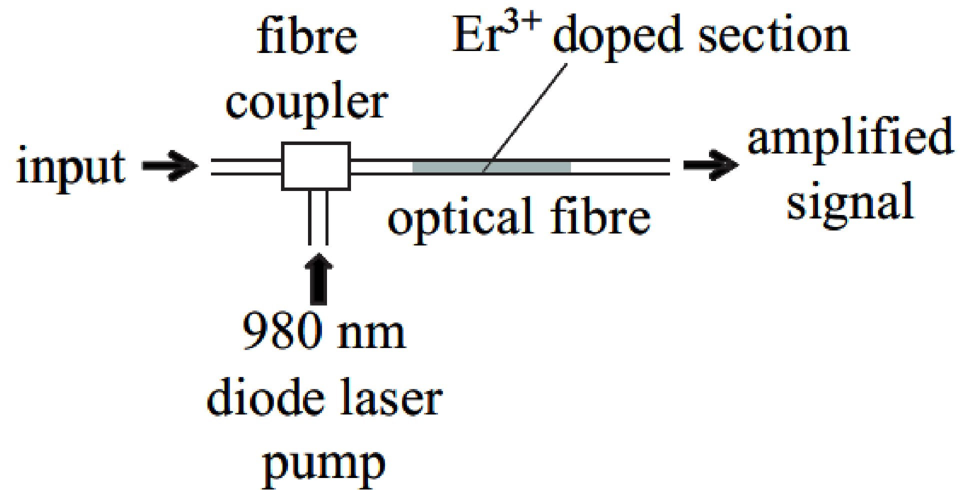
# Erbium fiber laser




3+ ion ground state: $^4I_{15/2}$

$Er^{3+}$: $4f^{11}$ $6s^0$ S=3/2, L=6, J=15/2




Pumped with 980 nm diode laser.
Important for telecommunications.

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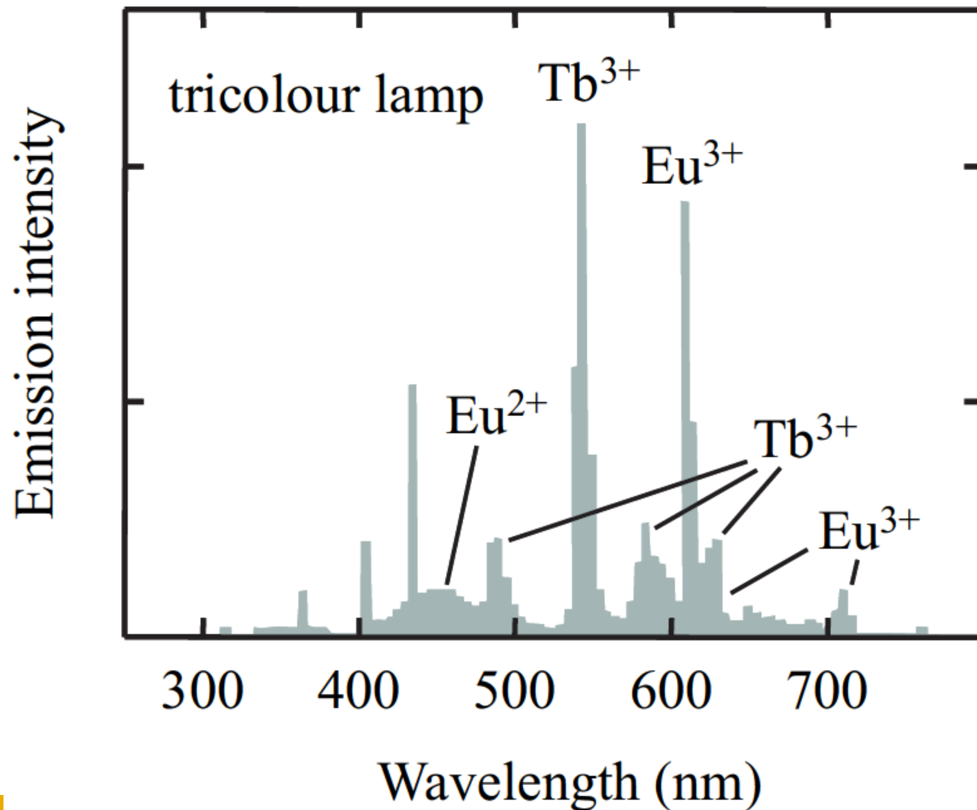
# Phosphors

Obsolete: Cathode ray tubes (TV, oscilloscope)
Fluorescent tubes.
Convert discrete LED (or Hg) emission into white light.




$BaMgAl_{10}O_{17}:Eu^{2+}$ ($4f^7$)
450 nm (blue)

$CeMgAl_{11}O_{19}:Tb^{3+}$ ($4f^8$)
550 nm (green)

$Y_2O_3:Eu^{3+}$ ($4f^6$)
610 nm (red)

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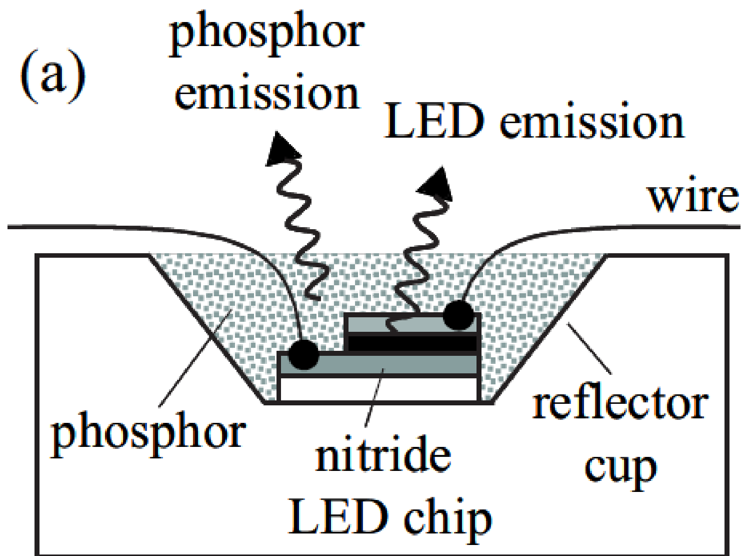
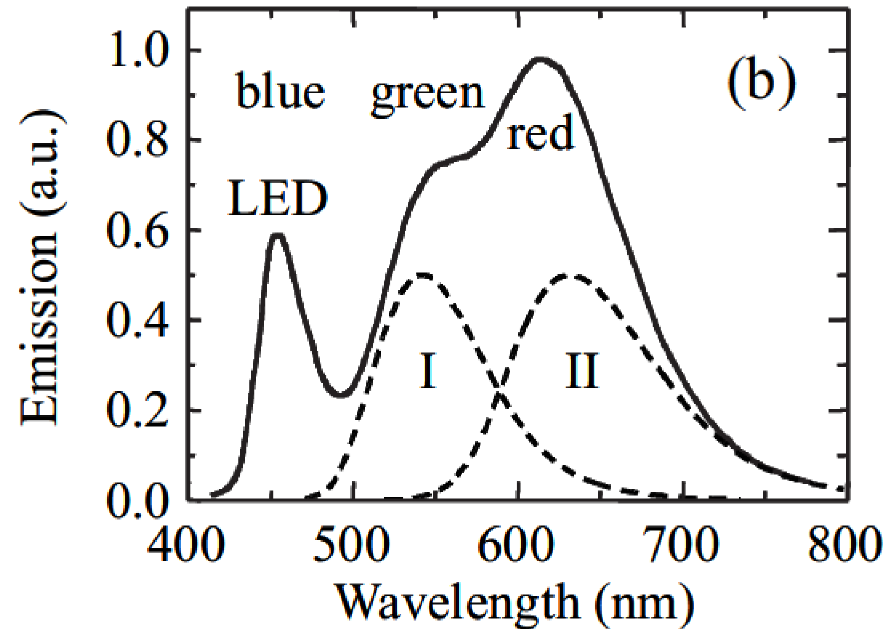
# White light emitting diodes (white LEDs)




Blue InGaN LED with green and red phosphors

I: $SrSi_2O_2N_2:Eu^{2+}$

II: $Sr_2Si_5N_8:Eu^{2+}$

Color temperature: 3200 K

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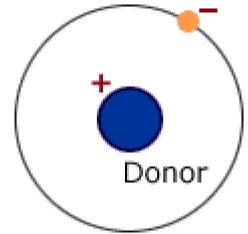
# Shallow (hydrogenic) defect: Si donor in GaAs

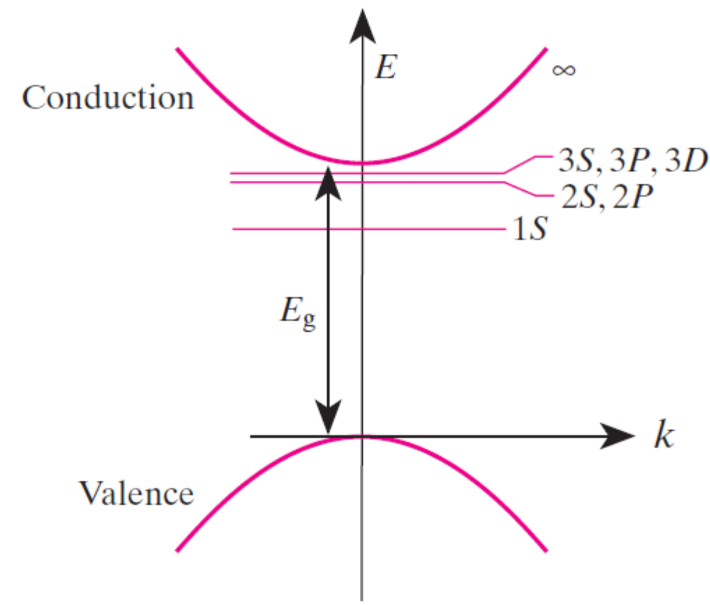



Extra valence electron
P nucleus positively charged (but screened in crystal)
Hydrogen-like energy spectrum (screened, heavy)

Extra potential: $V_S = +\frac{|e|}{4\pi\varepsilon_0 r}$

Rydberg series: $E = E_{CBM} - \frac{R}{n^2}$

Binding energy: $R = \frac{m^*}{m_0}\frac{1}{\varepsilon_S^2}\frac{e^4 m_0}{2\hbar^2(4\pi\varepsilon_0)^2}$




Bohr radius.
Similar to exciton problem.

Yu & Cardona

# Shallow (hydrogenic) defect: Si donor in GaAs

| Semiconductor | Binding energy from (4.24) [meV] | Experimental binding energy of common donors [meV] |
|---|---|---|
| GaAs | 5.72 | $Si_{Ga}$(5.84); $Ge_{Ga}$(5.88)<br>$S_{As}$(5.87); $Se_{As}$(5.79) |
| InP | 7.14 | 7.14 |
| InSb | 0.6 | $Te_{Sb}$(0.6) |
| CdTe | 11.6 | $In_{Cd}$(14); $Al_{Cd}$(14) |
| ZnSe | 25.7 | $Al_{Zn}$(26.3); $Ga_{Zn}$(27.9)<br>$F_{Se}$(29.3); $Cl_{Se}$(26.9) |

Works quite well for s-like conduction bands.
Complications due to p-like VB, anisotropic bands (Ge, Si, GaP)

Yu & Cardona

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# Shallow (hydrogenic) defect: Donors in Si and Ge

Theory | Li | P | As | Sb | Bi | S

Energy [meV]

0.0, –10.0, –20.0, –30.0, –40.0, –50.0

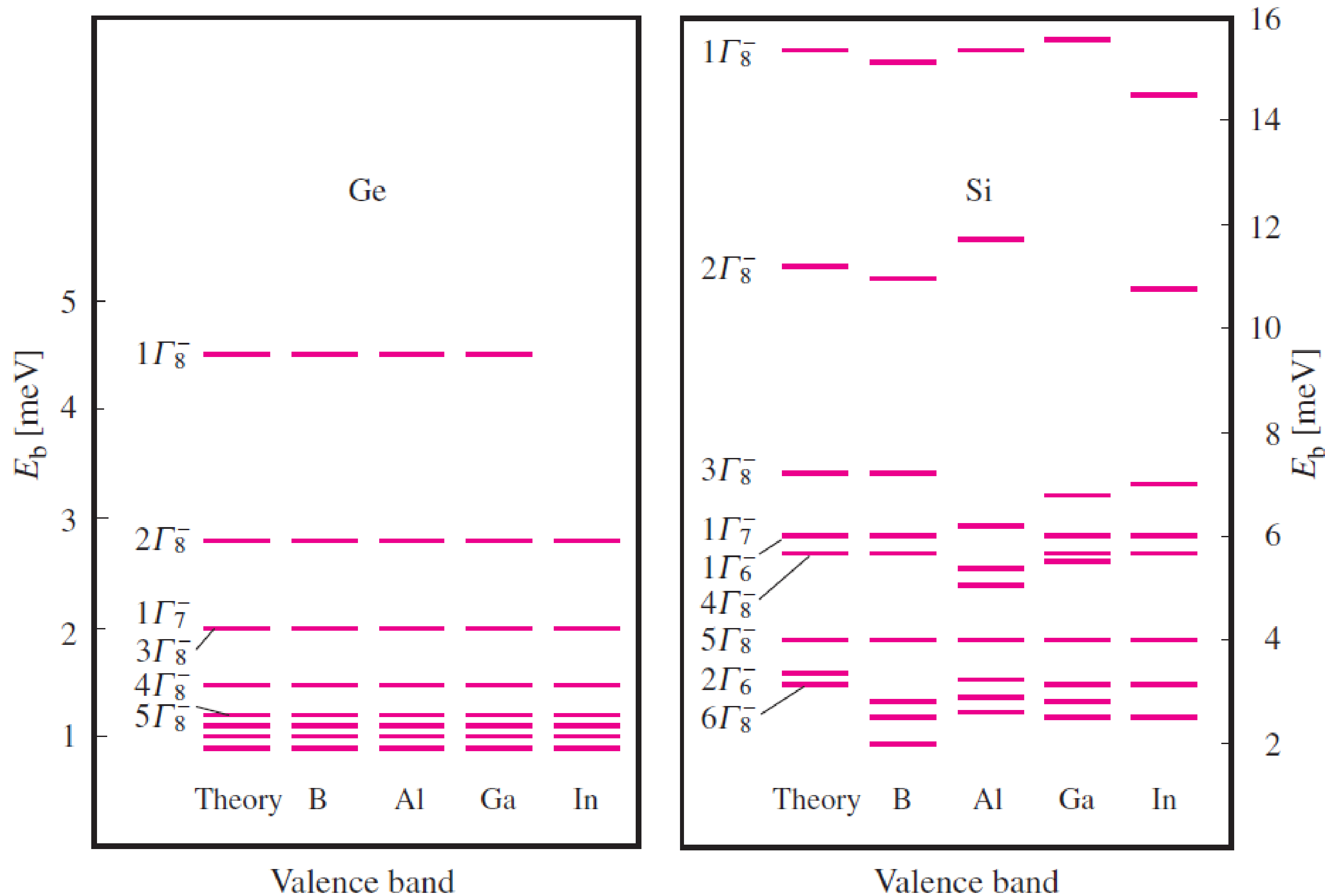
$4F_\pm$, $5F_\pm$, $6P_0$, $5P_\pm$, $6P_\pm$, $6H_0$, $5F_\pm$, $4F_0$, $4P_0$, $3P_\pm$, $6F_0$, $4P_\pm$, $3P_0$, $2P_\pm$, $5P_0$, $2P_0$

$1S$, $A_1$, $E+T_2$, $E$, $T_2$, $E$, $T_2$, $E$, $>T_2$, $A_1$, $A_1$, $A_1$

Poor agreement with theory (valley-orbit coupling)

Yu & Cardona

# Shallow (hydrogenic) defect: Acceptors in Si and Ge




Need to include valence band warping (Luttinger theory) and other corrections.

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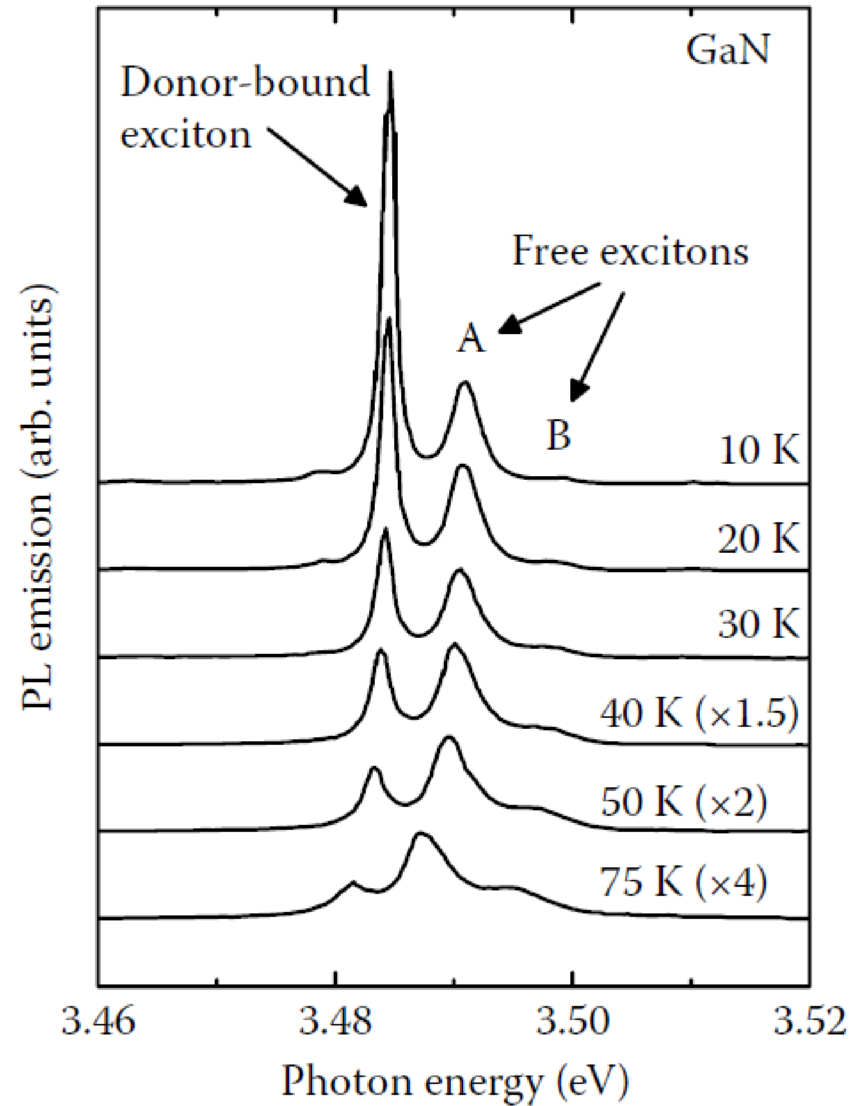
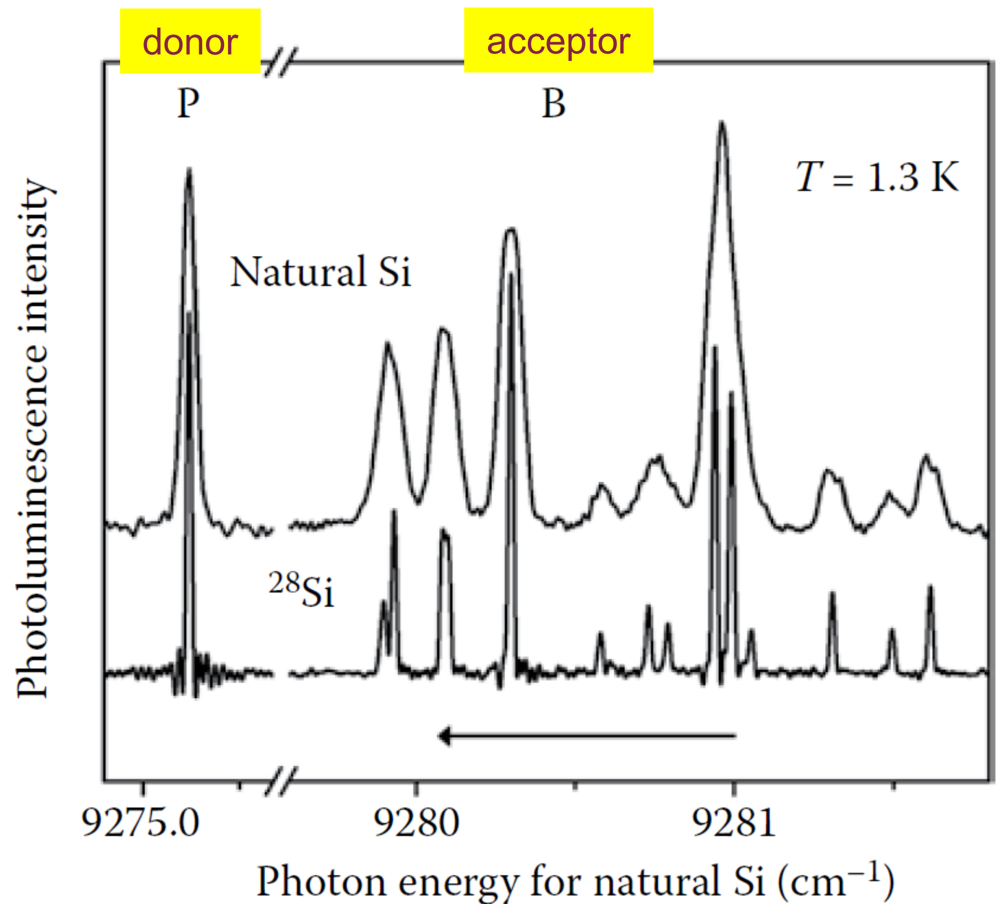
# Free and bound excitons (photoluminescence)




Band to defect recombination shifted from free exciton energy.

McClusky & Haller, Dopants and Defects

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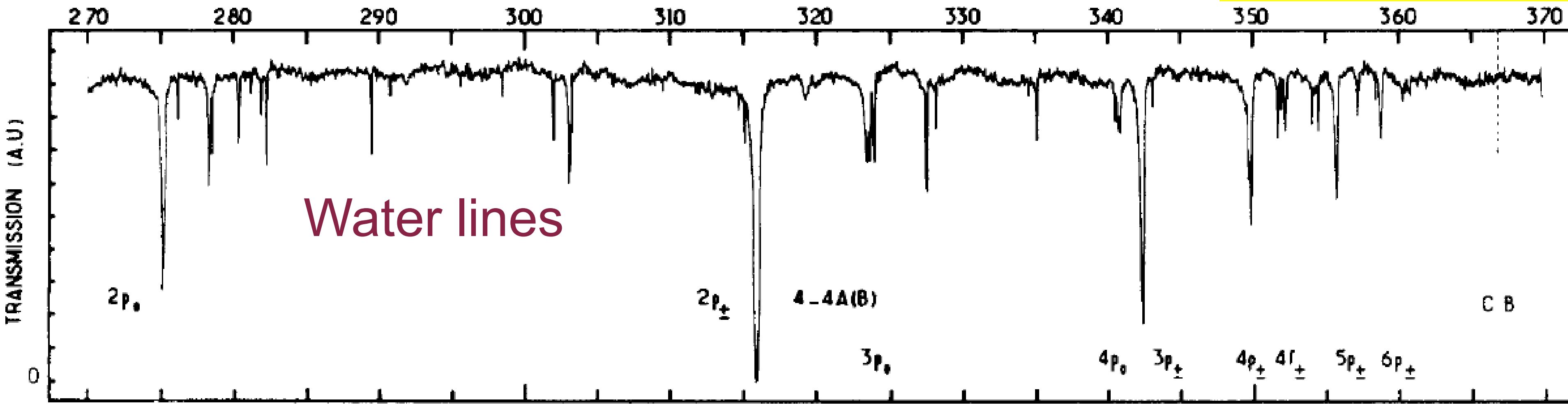
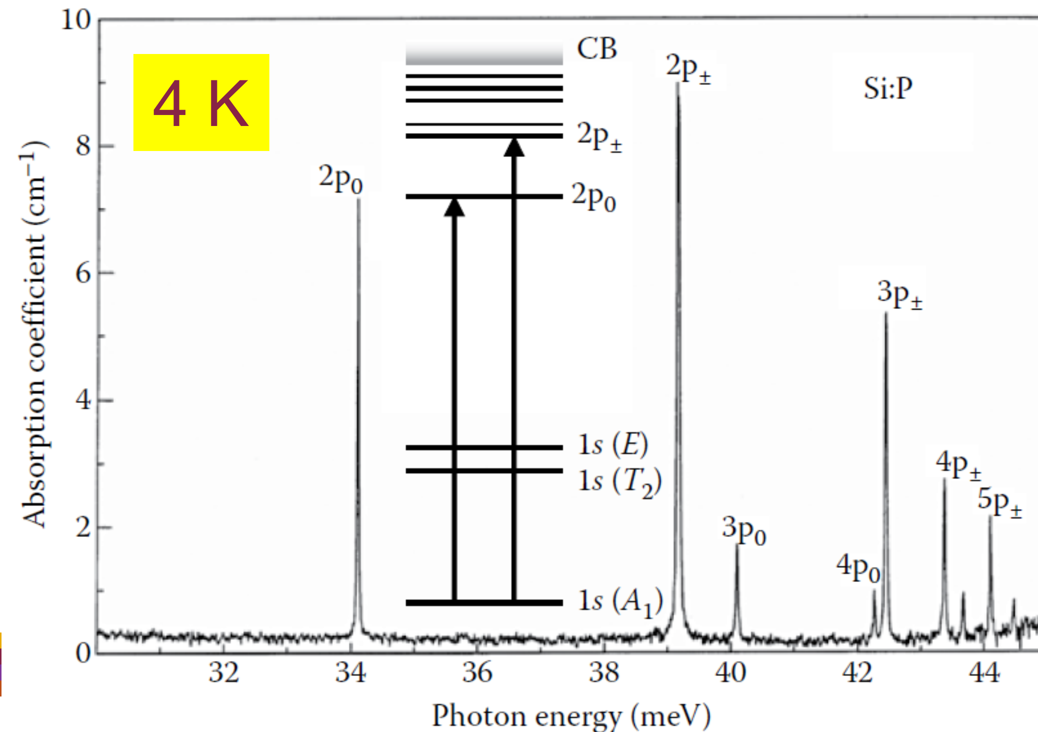
# Infrared defect spectroscopy: P donor in Si

Energy ($cm^{-1}$)



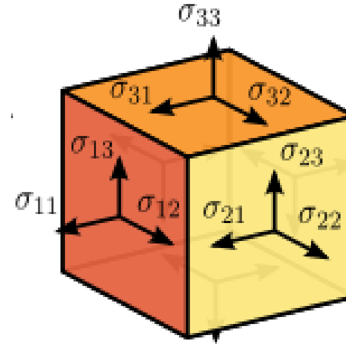



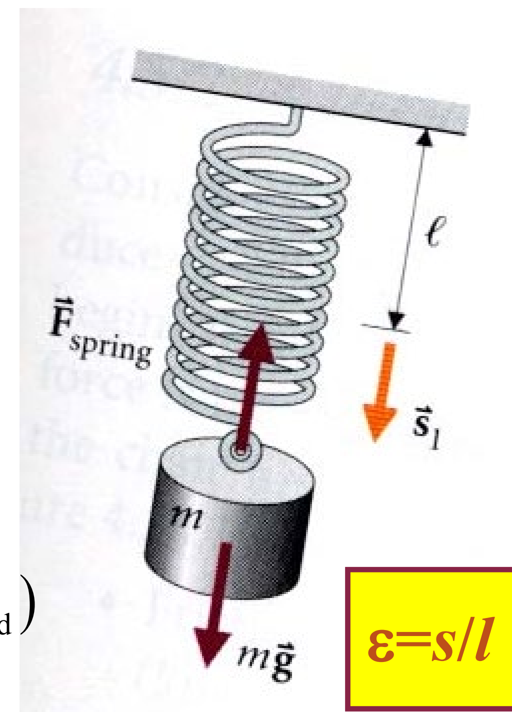

McClusky & Haller, Dopants and Defects
A.K. Ramdas, PRB **23**, 2082 (1981)
Pajot, Solid State Commun. **31**, 759 (1979)

# Stress and Strain: Hooke's Law

- Stress: Force per unit area (GPa), (1st rank tensor)

$$\mathbf{X} = \begin{pmatrix} X_{11} & X_{12} & X_{13} \\ X_{21} & X_{22} & X_{23} \\ X_{31} & X_{32} & X_{33} \end{pmatrix}$$




ε=s/l

- Strain: Describes the response to the stress, (1st rank tensor)

$$\varepsilon = \begin{pmatrix} \varepsilon_{11} & \varepsilon_{12} & \varepsilon_{13} \\ \varepsilon_{21} & \varepsilon_{22} & \varepsilon_{23} \\ \varepsilon_{31} & \varepsilon_{32} & \varepsilon_{33} \end{pmatrix} = \vec{\nabla} \otimes \left( \vec{a}_{\text{strained}} - \vec{a}_{\text{unstrained}} \right)$$

- Constituent relation (Hooke's law) between stress and strain: compliance tensor $\mathbf{S}$, stiffness tensor $\mathbf{c}$ (6x6 2nd rank tensors)

$$\varepsilon = \mathbf{S}\mathbf{X}$$

$$\mathbf{X} = \mathbf{c}\varepsilon$$

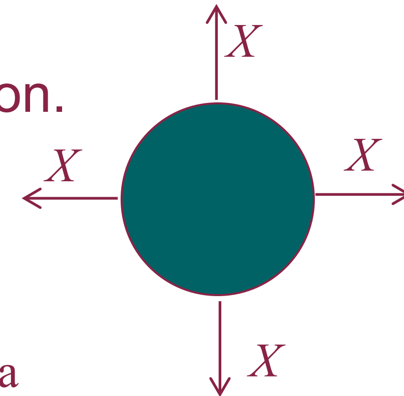
Spring: Strain is $\varepsilon = s/l$,
Relative change in length.

Landau & Lifshiftz, *Elasticity Theory*
Yu & Cardona, *Fundamentals of Semiconductors.*

# Biaxial Stress in Epitaxial Film on (001) substrate

- Stress:
  Forces act along the wafer, but not along the growth direction.
  - SiGe on Si: Compressive stress (X<0)
  - Si on SiGe: Tensile stress (X>0)

$$\mathbf{X}_{\text{biaxial}} = \begin{pmatrix} X & 0 & 0 \\ 0 & X & 0 \\ 0 & 0 & 0 \end{pmatrix}$$




- Resulting Strain:

$$\boldsymbol{\varepsilon}_{\text{biaxial}} = \begin{pmatrix} (S_{11}+S_{12})X & 0 & 0 \\ 0 & (S_{11}+S_{12})X & 0 \\ 0 & 0 & 2S_{12}X \end{pmatrix} = \begin{pmatrix} \varepsilon_{\|} & 0 & 0 \\ 0 & \varepsilon_{\|} & 0 \\ 0 & 0 & \varepsilon_{\perp} \end{pmatrix}$$

$\varepsilon_{\|} = 0.553\ X/10^{11}$ Pa
tensile in-plane strain

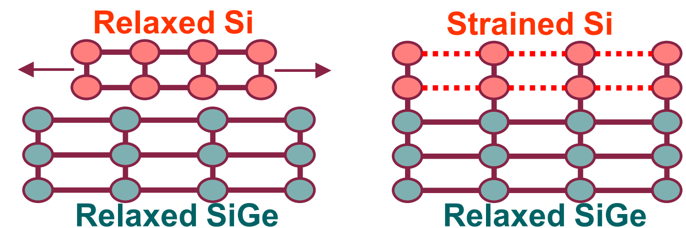
$\varepsilon_{\perp} = -0.979\ X/10^{11}$ Pa
compressive vertical strain

- Hydrostatic strain component: tensile (>0), softens phonons, gap decreases

$$\varepsilon_{\text{H}} = \left(2\varepsilon_{\perp} + \varepsilon_{||}\right)/3$$

- (100) Shear strain component: compressive (<0),
  Splits phonons and bands into singlet/doublet. Selection rules!

$$\varepsilon_{\text{S}} = \left(\varepsilon_{||} - \varepsilon_{\perp}\right)/3$$




S. Zollner, *Properties of Silicon-Germanium-Carbon Alloys: Growth, Properties, Applications*



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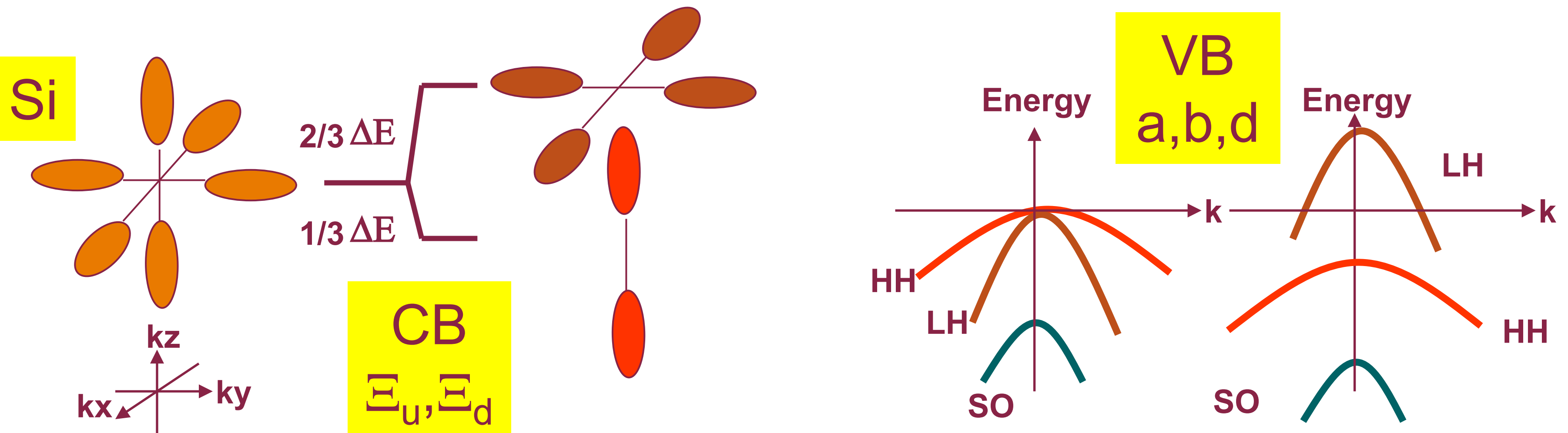
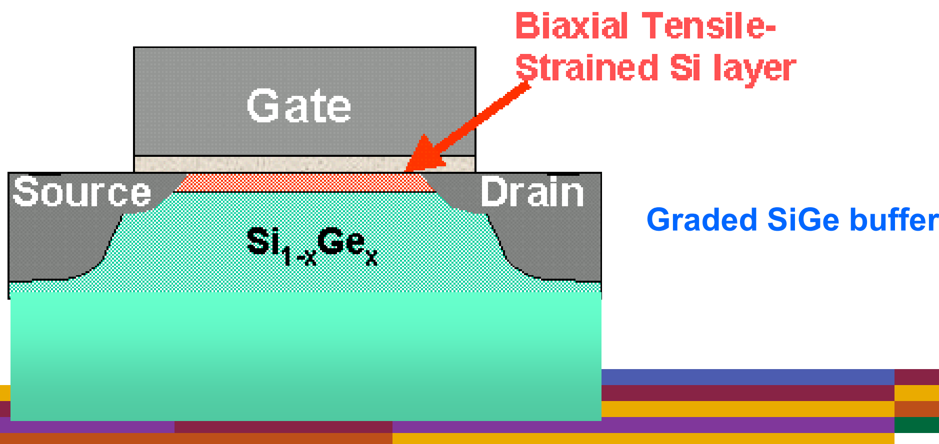
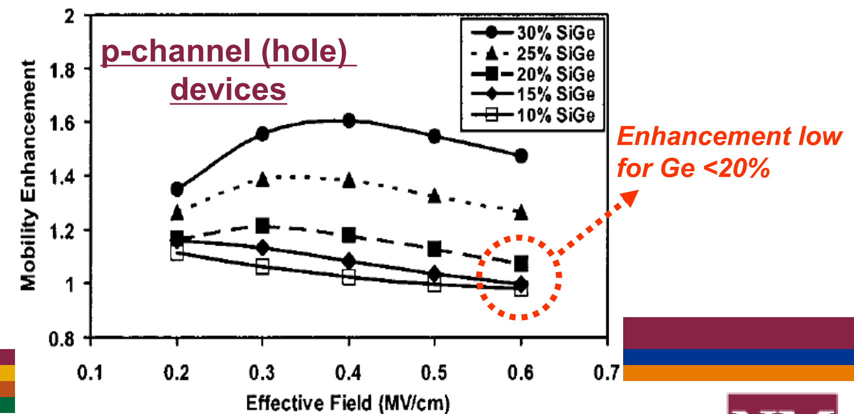
# Thin Strained Silicon Layers for CMOS




- Biaxial tension lowers band gaps, reduces effective masses, and splits bands/valleys (reduced intervalley and inter-valence band scattering).
- Silicon under biaxial tension has higher electron and hole mobilities and therefore offers better transport properties than bulk silicon.



# Deformation potentials

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**Deformation Potentials and Mobilities in Non-Polar Crystals**

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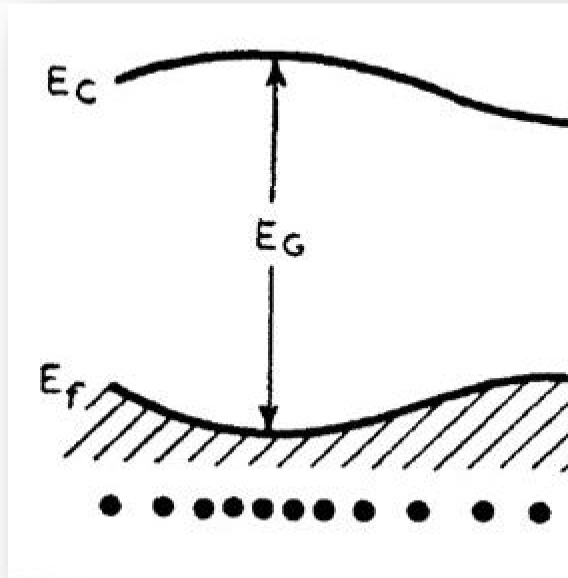
$$a = \frac{1}{V}\frac{\partial E_g}{\partial V}$$

$$\Delta E_g = a\,\mathrm{Tr}(\varepsilon)$$




$$a_V^{\Gamma\text{-}\Gamma} = \frac{\partial E_g^{\Gamma\text{-}\Gamma}}{\partial \ln V}$$

$$a_P^{\Gamma\text{-}\Gamma} = -\left(\frac{1}{B}\right)a_V^{\Gamma\text{-}\Gamma}$$

# Summary

- Quantum confinement and Heisenberg uncertainty principle
- Growth of quantum structures
- **Carbon nanostructures, two-dimensional materials**
- **Electronic states, quantum well absorption and emission**
- **Intersubband transitions**
- **Metamaterials and metasurfaces**

- **Defects**
- **Transition metal and rare earth impurities in insulators**
- **Shallow defects in semiconductors**

- **Stress and strain, deformation potentials**

NM STATE